# Fabrication and Characterization of Humidity Sensors based on CVD Grown MoS<sub>2</sub> Thin Film

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Abstract— Recent advances in two-dimensional (2D) transition metal dichalcogenides have demonstrated their potential application in chemical sensors. However, the chemical vapor deposition (CVD) grown molybdenum disulfide (MoS<sub>2</sub>) humidity sensors are still largely unexplored. In this work, MoS<sub>2</sub> thin films were grown on 1 cm<sup>2</sup> sapphire substrates through sulfurization of e-beam deposited Mo layers. The MoS<sub>2</sub> film morphology, thickness, and crystallinity were characterized by AFM and Raman spectroscopy. The twoterminal devices were fabricated with e-beam evaporated interdigitated electrodes (IDEs) on top of the MoS<sub>2</sub> surface. The water vapor sensing was tested at various humidity levels with the observed increase in the device resistance response to humidity due to the charge transfer mechanism. We found the devices to be reproducible and with excellent dynamic hysteresis. The sensitivity, fast response and recovery proved that CVD growth MoS<sub>2</sub> thin film could be scaled up for humidity and gas sensing applications.

#### I. INTRODUCTION

Two dimensional (2D) materials such as graphene and molybdenum disulfide (MoS<sub>2</sub>), have been extensively studied because of the unique electrical and optical properties introduced by their atomically-thin layer structure. [1]-[3] Various applications such as graphene integrated circuit [3], graphene field-effect transistors (FETs) [4] and MoS<sub>2</sub> photodetectors [5] are reported. Recent studies has shown chemical sensing applications with excellent performance of graphene [6] and graphene oxide [7].

Transition metal dichalcogenide (TMDC) materials are now drawing considerable attention due to their tunable bandgap and low background carrier densities.

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These properties can potentially provide high sensitivity of the sensors and high selectivity to different analytes. [8]-[11] In particular, sensing capabilities of  $MOS_2$  have recently been studied. Most of the reported sensors are based on mechanically exfoliated flakes of TMDC materials. [12], [13] For example, Late *et al.* fabricated few-layer exfoliated  $MOS_2$  FETs and demonstrated their sensitivity to NH<sub>3</sub>, NO<sub>2</sub> and humidity. [14] However, CVD deposited thin films have distinct advantages such as wafer-scale growth and compatibility with conventional semiconductor fabrication processes. [15]

To date, only Liu *et al.* examined the  $NH_3$  and  $NO_2$  gas sensing ability of CVD grown ultra-thin  $MoS_2$  films. [16] In addition, Liu *et al.* reported humidity sensing on the devices fabricated using physical vapor deposition (PVD) of thick  $MoS_2$  polycrystalline films on Si substrate. [17]

To explore the sensing properties of the CVD-grown  $MoS_2$  thin films, we studied two-terminal sensors, which show a resistance change in response to the absorption of water molecules. We measured the current-voltage curves at different humidity levels and recorded a positive response of increased resistivity to higher humidity. Reproducibility and dynamic hysteresis were tested, which proved the promising humidity sensing capacity of ultra-thin  $MoS_2$  films.

# II. EXPERIMENTAL

# A. Thin Film Growth and Device Fabrication

In this study, we built a  $MoS_2$  thin film sensor prepared by sulfurization of deposited Mo layer. [18] A thin layer of Mo, 4 nm thick, was deposited onto RCA cleaned c-axis sapphire substrates by e-beam deposition.  $MoS_2$  films were then produced by sulfurization of Mo in an H<sub>2</sub>S/H<sub>2</sub>/Ar gas mixture using a custom-build CVD reactor. The chamber temperature and pressure was set to 1000 °C and 1.33 kPa, respectively. After 20 min of sulfurization, substrates were rapidly cooled to  $\approx$ 500 °C to prevent re-evaporation of the synthesized material.

Ti (40 nm)/Au (160 nm) bi-layer comb-like IDEs were ebeam deposited on lithographically patterned MoS<sub>2</sub> followed by lift-off. The electrode structure configuration is as follows: the length of each finger is 1200  $\mu$ m with the gap of 100  $\mu$ m. The corner contact pads size is 400  $\mu$ m × 400  $\mu$ m and the full scale of the device is 2 mm × 2 mm. SEM image of the fabricated device is shown in Figure 1.

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Fig. 1. SEM image of the fabricated structure

The Ti/Au metal contact combination helps narrow the potential barrier at the interface and maximize carrier injection. [19] In general, this structure is shown to provide higher current which improves signal-to-noise ratio and hence sensitivity.

## B. Characterization and Measurements

Atomic Force Microscopy (AFM) image (Figure 2(b)) reveals hexagonal grains of  $MoS_2$  film most of which have the same orientation. This is indicative of quasi-epitaxial growth of  $MoS_2$  on sapphire. The thickness of the film was 9 nm (Figure 2(a)). For different samples, surface roughness is estimated between 3.0 nm and 4.2 nm. A representative Raman spectrum (Laser Quantum DPSS, 532 nm) [20] exhibits characteristic lines corresponding to the  $E_{2g}$  and  $A_{1g}$  modes of  $MoS_2$  with the full width at half maximum (FWHM) between 3.1 cm<sup>-1</sup> and 3.6 cm<sup>-1</sup> (Figure 2(c)). Small FWHM values are usually correlated with a high structural quality of the material.



Fig. 2. (a) Low- and (b) high-resolultion AFM images of  $MoS_2$  film. Inset in (a) shows an AFM height profile of  $MoS_2$  film. (c) Roomtemperature Raman spectrum of  $MoS_2$  film produced by Mo sulfurization. Raman peak at 418 cm<sup>-1</sup> marked with an asterisk originates from the sapphire substrate.

Current response to changes in the relative humidity (RH) was recorded at a 2 V bias using a NI PXI-1033 instrument. RH was varied by diluting humid air flow (RH = 40 %) with dry air so that 35 %, 30 %, and 25 % RH values were obtained using 20 sccm, 60 sccm, and 100 sccm dry air flow, respectively. The RH was measured by a fast-response commercial humidity probe placed at the exhaust of the sensing chamber. The sensitivity was defined as

$$\mathbf{S} = (I_{RH1} - I_{RHE})/I_{RHE},\tag{1}$$

where  $I_{RH1}$  is the current of the device at a certain RH value and  $I_{RHE}$  is the current at the base humidity (40 %).

### III. RESULTS AND DISCUSSION

Dynamic humidity response curves of a  $MoS_2$  thin film sensor under the base humidity and 25 % RH plotted in Figure 3(a) show that current increases for lower RH. A sensitivity of 5.5 % was calculated. We tested the response to dry air and dry N<sub>2</sub> to probe the impact of oxygen. From Figure 3 (b) we can see a 0.3 % change for the two gases, which indicates that the influence of oxygen is much smaller than the humidity. From Figure 4 (b), a response/recovery time of 250 s/250 s was also measured. This parameter is higher compared to some other reported  $MoS_2$  humidity sensors.[11]



Fig. 3. (a) Typical current-voltage (IV) curves obtained at room humidity and 25 % RH (b) response cycle to dry air and dry N<sub>2</sub>.

The reproducibility of the device is shown in Figure 4(a). In this test, the device was exposed to a step-like increased flow of dry air with an exposure time of 500 s for each RH level to reach a current saturation. The test result demonstrated a remarkable reproducibility with three different humidity levels and only a small baseline drift of 0.3 % is observed for each cycle. The sensitivity curve of humidity from 25 % RH to 40 % RH is demonstrated in

Figure 4(b), from which we can conclude that the device performance is better for lower humidity values.



Fig. 4. (a) The reproducibility of the device at 25 % RH, 30 % RH and 35 % RH (b) The sensititivity curve of humidity from 25 % RH to 40 % RH

Humidity sensors are often affected by the history of its environment conditions, and the dynamic hysteresis is also one of the most important figures of merit of the humidity sensors. The humidity hysteresis of MoS<sub>2</sub> sensor was tested by increasing the dry air flow rate from 20 sccm to 120 sccm at a 20 sccm increment and then decreasing to 80 sccm and further to 40 sccm. The dynamic humidity hysteresis curve is plotted in Figure 6 with a hysteresis about 1 % RH, which is better than for some other 2D materials humidity sensors. Wang et al. proposed that the hysteresis in I-V characteristics is caused by the charge injection at the surface and the charge transfer between neighboring adsorbates. [21]



Fig. 5. Dynamic hysteresis curve of the device

In general, the positive impedance response of  $MoS_2$  thin film humidity sensor can be explained according to the charge transfer concept: the surface absorbed water molecules act as electron acceptors, resulting in pinning of the Fermi level closer to the valence-band edge. [12], [22] The free electrons in  $MoS_2$  interact with the water molecules according to the reaction:

$$H_2 0 + e^- = \frac{1}{2} H_2 + 0 H^-.$$
 (2)

In this way, the free electrons are consumed which leads to the decrease of the current. From the mechanism described above, we can speculate that the sensitivity might be influenced by the thickness of the sample and by the operating conditions, for example the surface oxidation of the film could cause a drift of the device.

#### IV. CONCLUSIONS

In this work, humidity sensors fabricated using large-area CVD-grown MoS<sub>2</sub> thin films are presented. A positive resistance response of the sensors to the humidity was observed with a high sensitivity ( $\approx 5.5$  % for a 15 % RH difference), a low hysteresis ( $\approx 1$  %), and response and recovery time of  $\approx 250$  s. The sensing mechanism is explained by electron transfer from MoS<sub>2</sub> to the water molecules, which results in a decrease of current with increasing humidity.

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