The relationship between local order, long range order, and sub-band-gap defects in hafnium oxide and hafnium silicate films

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We have measured x-ray absorption spectra (XAS) at the oxygen *K* edge for hafnium oxide (HfO₂) films grown by chemical vapor deposition (CVD) and atomic layer deposition (ALD), as well as for hafnium silicate (HfSiO) films grown by CVD. The XAS results are compared to x-ray diffraction (XRD) and spectroscopic ellipsometry (SE) data from the same films. Features characteristic of crystalline HfO₂ are observed in the XAS spectra from all CVD-grown HfO₂ films, even for a thickness of 5 nm where XRD is not sensitive. XAS and XRD spectra from the ALD-grown HfO₂ films exhibit the signature of crystallinity only for films that are 20 nm or thicker. These characteristic XAS features are absent in all HfSiO films measured, which is consistent with their being amorphous. The appearance of these peaks in XAS and XRD is correlated with sub-band-gap absorption in the SE spectra, which appears to be intrinsic to crystalline HfO₂ in the monoclinic phase. © 2008 American Institute of Physics. [DOI: 10.1063/1.2909442]

The historic increase in semiconductor logic integration relies on exponential downscaling in the size of individual metal-oxide-semiconductor field-effect transistors. As the capacitance of the gate insulator stack needs to be increased simultaneously with a reduction in channel length to maintain gate control over the inversion layer, this has led to a reduction in SiO₂ or SiON gate insulator thickness to ~ 1 nm in high-performance logic chips. Despite the nearly perfect electrical properties of the SiO₂/Si interface, at subnanometer thicknesses, electron tunneling through the oxide results in an unacceptably high gate leakage current even at modest biases. This has led to a large body of research aimed at replacing SiO_2 with a high-permittivity (high-K) dielectric which could provide the needed capacitance for a thicker film, thus mitigating the leakage current problem.¹⁻⁴ Hafnium oxide (HfO₂) and hafnium silicates (HfSiO) are promising candidate high-K dielectrics. However, questions remain regarding the impact of structural properties such as crystallinity on electrical performance. Crystallinity of Hfbased dielectrics depends on factors such as composition, thickness, deposition conditions, and thermal budget during device integration which usually involves annealing at 1000 °C or more. A number of reasons, often speculative, have been put forward for aiming to keep high-K dielectrics amorphous: (a) heterogeneous grain orientations could give rise to spatially varying electric fields in the channel, causing carrier scattering and degrading mobility. (b) Grains could cause device-to-device variations in, e.g., leakage or threshold voltage. (c) Grains could increase effective line edge roughness after gate etch. (d) Grain boundaries have been claimed to be responsible for localized unoccupied gap states,^{5–7} e.g., those observed $\sim 0.2-0.3$ eV below the conduction band edge if and only if HfO₂ is crystalline;⁸ such gap states may explain undesirable electron trapping and gate leakage with HfO₂.^{9,10}

In this work, we report soft x-ray absorption spectroscopy (XAS) measurements performed on a series of HfO₂ and HfSiO films grown by metal organic chemical vapor deposition (MOCVD) and HfO₂ films grown by atomic layer deposition (ALD) to probe the local bonding and short-range order in these systems. Characteristic features associated with crystalline HfO₂ are observed in the XAS spectra from films that also exhibit well-defined peaks, associated with the monoclinic phase, in XRD. These XAS features are even observed for 5 nm thick CVD films where crystallites are too small to be detected by XRD. By comparing the XAS and XRD results with the optical absorption data, we find that the HfO₂ crystalline state and a sub-band-gap optical absorption feature associated with electronic defects, coincide in all cases. Our results suggest that the so-called defect feature need not be attributed to structural distortions of nanocrystalline (~ 1 nm) particles or to structural modifications at grain boundaries in the films, as has been proposed.⁵⁻⁷ Rather, these features appear to be intrinsic to the formation of monoclinic HfO₂ crystallites in the films. Our work thus contributes to a fundamental understanding of the electronic structure of HfO₂ and its electrical properties as a candidate gate dielectric.

Measurements were performed on three sets of films. These sets were composed of MOCVD-grown HfO₂ and HfSiO [Hf:Si~80:20] samples, as well as a set of ALDgrown HfO₂ films. Each set contained films of 5, 10, 20, and 40 nm of nominal thicknesses. The MOCVD HfO₂ films were grown using hafnium tetra-*tert*-butoxide [HTB, Hf(OC(CH₃)₃)₄] and O₂. To grow the HfSiO films, SiH₄ was

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FIG. 1. (Color online) X-ray absorption spectroscopy (XAS) data of the O K-edge from HfO₂ films deposited by (a) MOCVD and (b) ALD, as well as (c) MOCVD-grown HfSiO films, for thicknesses of 5, 10, 20, and 40 nm.

added to the precursor mix. The ALD films were deposited via alternating exposures of HfCl₄ and H₂O using N₂ as a carrier gas. The films were deposited on highly doped 200 mm Si(100) wafers (*n* type, ~1 Ω cm, 800 μ m thick) whose surfaces were cleaned by an SC-1/SC-2/HF process followed by thermal oxidation using NO which resulted in the formation of an 11 Å oxynitride interface layer prior to deposition of the high-*K* films.

X-ray absorption spectroscopy (XAS) measurements were performed on the U4A beamline of the National Synchrotron Light Source at Brookhaven National Laboratory. To acquire the O K-edge spectra, the photon energy was scanned from 500 to 600 eV in 0.1 eV increments while the total electron yield was measured by monitoring the total photocurrent emitted from the sample. The total energy resolution was 0.1 eV and all data were normalized to the incident flux. X-ray diffraction (XRD) spectra were obtained from a Bruker-AXS D8 general area detector diffraction system with Cu K α radiation. Two dimensional 2θ - χ patterns were collected in the 2θ range from 15° to 48° and integrated in χ to obtain the patterns. The spectroscopic ellipsometry measurements were performed on a vacuum ultraviolet spectroscopic ellipsometer. The dielectric functions $\varepsilon = \varepsilon_1 + i\varepsilon_2$, where ε_1 and ε_2 are the real and imaginary parts of ε , respectively, were determined from the analysis of the ellipsometric data.

Figures 1(a)–1(c) show a series of XAS spectra for the O *K*-edge obtained from MOCVD-grown HfO₂, ALD-grown HfO₂, and MOCVD-grown HfSiO films, respectively, with thicknesses of 5, 10, 20, and 40 nm. All of the spectra show a similar overall shape composed of two main features, one at ~532.3 eV and the other at ~536.5 eV, followed by a gradually decreasing intensity to ~550 eV, above which the intensity is almost constant. These two main peaks are associated with transitions to unoccupied O 2*p* levels that are hybridized with Hf 5*d* orbitals. The lower energy peak (~532.3 eV) and the higher energy peak (~536.5 eV) are the crystal-field-split e_g and t_{2g} components, respectively, of the Hf 5*d* manifold.^{11,12} Upon closer inspection, however, it

is clear that for all film thicknesses, the spectra from MOCVD-grown HfO₂ films exhibit a significant amount of fine structure that is absent in the spectra from the HfSiO films. In particular, the spectra from the MOCVD-grown HfO₂ films have a series of three well-defined peaks at 540, 542, and 548.5 eV. Emission in this energy range is associated with excitation to O 2*p* levels hybridized with Hf 4*s* and 4*p* states and it is well established that the appearance of these three peaks is a signature of crystalline HfO₂.^{13–18} Another important observation is that the e_g -related feature of the MOCVD-grown HfO₂ films is split into two components at 532.0 and 532.9 eV. In the spectra from the HfSiO films, the absence of this splitting, as well as the lack of the three peaks at higher energies, is consistent with the HfSiO being amorphous for all thicknesses examined here.

In contrast to the MOCVD-grown films of either composition, the XAS spectra from the ALD-grown HfO₂ films change as a function of film thickness. The spectra from the 5 and 10 nm films strongly resemble those from the HfSiO films. The shoulder at 540 eV shows slightly more definition, but the characteristic three peak pattern is absent. On the other hand, the spectra for the 20 and 40 nm films do exhibit the three higher energy peaks, essentially identical to what is observed in the spectra from the MOCVD-grown HfO₂ films. The similarity of these spectra is even more compelling in Fig. 2 where spectra from the 5 and 40 nm films of each type are presented on the same plot. While the relative intensity of the fine structure may slightly vary with film thickness, possibly owing to changing surface-to-volume ratio of the crystallites or (in the ALD case) the presence of a second minor phase, the energies of each feature in the thicker ALD-grown films and the MOCVD-grown HfO₂ films are identical indicating that the ALD-grown films are crystalline for thicknesses ≥ 20 nm.

To investigate the relationship between local bonding, as probed by XAS, and long range order as probed by x-ray diffraction, we obtained XRD data from the MOCVD-grown and ALD-grown HfO₂ films, which are presented in Figs. 3(a) and 3(b), respectively. Each plot contains data from



FIG. 2. (Color online) Comparison of selected XAS spectra from Fig. 1.

films with nominal thicknesses of 5, 10, 20, and 40 nm. As evident from Fig. 3(a), well-defined diffraction peaks are present in the data from the 10, 20, and 40 nm MOCVDgrown films. No definitive diffraction features are observed in the 5 nm data due to limited XRD sensitivity. Indexing the peaks for the thicker films shows that the crystallites are in the monoclinic phase. For the ALD-grown films, Fig. 3(b) shows no diffraction features for the 5 and 10 nm thicknesses. The data from the 20 nm film show weak diffraction features while the 40 nm film exhibits well-defined diffraction peaks. Indexing these XRD data shows that most of the peaks are associated with the monoclinic phase, although several weak features suggest the presence of small amounts of material in the orthorhombic or tetragonal phases as well. XRD data obtained from the four HfSiO films [Fig. 3(c)] contain no evidence of diffraction features, indicating that under these deposition and processing conditions, this material remains amorphous up to thicknesses of 40 nm or more.

Figures 4(a)–4(c) summarize the ellipsometric data from CVD-grown HfO₂, ALD-grown HfO₂, and CVD-grown Hf-SiO films, respectively. For all of the HfSiO films, there is an onset at ~5.75 eV corresponding to absorption across the band gap of the silicate. In contrast, the data from the CVD-grown HfO₂ films show a pronounced feature at lower energies, with an edge at 5.5 eV that plateaus, followed by a rapid rise above 6 eV. The rapid rise is associated with absorption across the band gap of HfO₂, while the former feature is attributed to defects ~0.2–0.3 eV below the optical band gap.⁸ These defects may be the cause of electron trap



FIG. 3. (Color online) X-ray diffraction (XRD) data from HfO_2 films deposited by (a) MOCVD and (b) ALD, as well as (c) MOCVD-grown HfSiO films. Extra red line in (b) is associated with either orthorhombic or tetragonal phase.

sites in device gate stacks observed at 0.25 and 0.35 eV below the HfO_2 conduction band edge.¹⁰ Although the shape of the HfO_2 absorption edge differs from that of HfSiO, for each material, the shape of the spectrum is the same for all film thicknesses. Figure 4(b) illustrates that this is not the case for the ALD-grown films. For the 5 and 10 nm films, the absorption edge is similar in shape to that of the HfSiO films (i.e., a single onset attributable to transitions across the band gap). In contrast, for the 20 nm film, additional absorption is seen below the energy of the band gap and the 40 nm film has a well-defined sub-band-gap feature that is essentially identical to that seen for the CVD-grown films.

By comparing the XAS data to the XRD and SE data, it is clear that in all cases, films that exhibit XRD peaks associated with monoclinic HfO_2 also show the characteristic three-peaked structure as well as the split peaks at 532.0 and 532.9 eV in XAS and a well-defined sub-band-gap absorption feature in SE. This correlation provides important new information about these material systems. It is well known that XAS spectra are highly sensitive to the local bonding environment of the species whose absorption edge is measured.^{11,12} This is primarily because of the participation



FIG. 4. (Color online) Spectroscopic ellipsometry (SE) data from HfO_2 films deposited by (a) MOCVD and (b) ALD, as well as (c) MOCVD-grown HfSiO films. For clarity, the 10, 20, and 40 nm curves are shifted upward by 1, 2, and 3 units, respectively.

in the absorption process of a core electron whose wave function is highly localized in the atomic site. The XAS data, therefore, reveal the presence of monoclinic HfO₂ nanocrystallites in the 20 nm ALD-grown HfO₂ films and the 5 nm MOCVD-grown films, which are undetected by the less sensitive XRD method. Moreover, there is a one-to-one correlation between the appearance of the monoclinic HfO_2 features in XAS and the presence of the "defect" feature in the optical data. The line shape of the XAS data for the 20 nm ALD film cannot be reproduced by a linear combination of the line shapes for the 40 nm ALD film plus either the 10 or 5 nm ALD film. This speaks against the interpretation that the 20 nm ALD HfO_2 film is composed of a collection of HfO_2 nanocrystallites embedded in an amorphous HfO2 matrix. Rather, it suggests that the film is dominated by small crystalline nuclei that have not yet been organized to establish long range order.

The splitting of the XAS features associated with the Hf e_g states (that is, the peaks at 532.0 and 532.9 eV) and the sub-band-gap SE absorption feature have attracted considerable attention in the literature. Recent studies have attributed both phenomena to a Jahn-Teller distortion of the octahedral oxygen cages surrounding Hf ions in nanocrystallites of HfO_2 or at grain boundaries.^{5–7} However, the thickness evolution in the data from our films suggests that both phenomena may instead be intrinsic to monoclinic HfO₂: for MOCVD-grown films, the XAS peak splitting is progressively more well defined as the film thickness increases, indicating increasing crystallite size and therefore a decreasing density of grain boundaries; however, the fraction of the SE absorption intensity attributed to the sub-band-gap feature remains almost constant. The discrepancy between these thickness trends appears to be inconsistent with an assignment to grain boundaries. This observation is similar to what was found in recent optical absorption experiments from sputter-deposited HfO2 films where increasing intensity of the sub-band-gap feature is correlated with increasing monoclinic crystallinity of the film.¹⁹ One could imagine that the sub-band-gap feature observed in our SE data could be due to O vacancies. However, the films studied here were deposited under oxidizing conditions and both the SE feature and the ~ 0.9 eV splitting in the XAS data are observed. Moreover, in a SE study of HfO_2 films deposited by ALD and annealed in N_2 ,²⁰ i.e., under reducing conditions that may be expected to increase the O vacancy concentration, no increase in sub-band-gap absorption intensity was reported. Cluster calculations modeling the XAS spectrum from O-deficient HfO_2 suggest that the near-edge splitting may be reduced rather than enhanced, for Hf ions near the O vacancy.²¹ Both of these observations make O vacancies an unlikely explanation for the sub-band-gap feature or the splitting in the XAS.

Therefore, we propose that the sub-band-gap feature and the splitting in XAS may be intrinsic properties of monoclinic HfO₂. Indeed, a recent cluster calculation of HfO₂ in the monoclinic structure shows that the near-edge region is dominated by the ²S and ²P configurations which form two well-defined features separated by ~1.25 eV, very similar to the 0.9 eV splitting we observe,²¹ and further supporting our proposal.

In summary, we have obtained O *K*-edge XAS spectra from 5, 10, 20, and 40 nm films of HfO₂ and HfSiO grown by CVD and HfO₂ films grown by ALD, and correlated the results with XRD and SE data from the same samples. All of the CVD-grown, as well as the two thickest ALD-grown, HfO₂ films are crystalline and show well-defined features in XAS that are absent from the amorphous samples. The XAS data indicate that monoclinic HfO₂ nanocrystallites are present in the 5 nm CVD-grown and the 20 nm ALD-grown HfO₂ films, even though XRD measurements are inconclusive. A splitting in the XAS spectra near threshold is correlated with sub-band-gap absorption in the SE spectra in all cases. The persistence of these features with increasing film thickness and improved crystallinity suggest that they are intrinsic to crystalline HfO₂ in the monoclinic phase.

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- ²*Materials Fundamentals of Gate Dielectrics*, edited by A. A. Demkov and A. Navrotsky (Springer, Dordrecht, 2005).
- ³J. Robertson, Rep. Prog. Phys. **69**, 327 (2006).
- ⁴G. D. Wilk, R. M. Wallace, and J. M. Anthony, J. Appl. Phys. **89**, 5243 (2001).
- ⁵G. Lucovsky, Appl. Surf. Sci. **253**, 311 (2006).
- ⁶G. Lucovsky, C. C. Fulton, Y. Zhang, Y. Zou, J. Luning, L. F. Edge, J. L.
- Whitten, R. J. Nemanich, H. Ade, D. G. Schlom, V. V. Afanase'v, A. Stesmans, S. Zollner, D. Triyoso, and B. R. Rogers, IEEE Trans. Device Mater. Reliab. 5, 65 (2005).
- ⁷G. Lucovsky, Y. Zhang, J. Luning, V. V. Afanase'v, A. Stesmans, S. Zollner, D. Triyoso, B. R. Rogers, and J. L. Whitten, Microelectron. Eng. **80**, 110 (2005).
- ⁸N. V. Nguyen, A. V. Davydov, D. Chandler-Horowitz, and M. M. Frank, Appl. Phys. Lett. **87**, 192903 (2005).
- ⁹X. Xu, M. Houssa, S. De Gendt, and M. Heyns, Appl. Phys. Lett. **80**, 1975 (2002).
- ¹⁰G. Bersuker, J. H. Sim, C. S. Park, C. D. Young, S. Nadkarni, R. Choi, and B. H. Lee, IEEE 44th Annual International Reliability Physics Symposium Proceedings, 179 (2006).
- ¹¹F. M. F. de Groot, M. Grioni, J. C. Fuggle, J. Ghijsen, G. A. Sawatzky, and H. Petersen, Phys. Rev. B 40, 5715 (1989).
- ¹²L. Soriano, M. Abbate, J. C. Fuggle, M. A. Jimenez, J. M. Sanz, C.

- Mythen, and H. A. Padmore, Solid State Commun. 87, 699 (1993).
- ¹³M. H. Cho, K. B. Chung, C. N. Whang, D. W. Lee, and D. H. Ko, Appl. Phys. Lett. 87, 242906 (2005).
- ¹⁴M. H. Cho, D. W. Moon, S. A. Park, Y. K. Kim, K. Jeong, S. K. Kang, D. H. Ko, S. J. Doh, J. H. Lee, and N. I. Lee, Appl. Phys. Lett. 84, 5243 (2004).
- ¹⁵H. Takahashi, J. Okabayashi, S. Toyoda, H. Kumigashira, M. Oshima, K. Ikeda, G. L. Liu, Z. Liu, and K. Usuda, Appl. Phys. Lett. 89, 012102 (2006).
- ¹⁶H. Takahashi, J. Okabayashi, S. Toyoda, H. Kumigashira, M. Oshima, K. Ikeda, G. L. Liu, Z. Liu, and K. Usuda, J. Appl. Phys. **99**, 113710 (2006).
- ¹⁷H. Takahashi, S. Toyoda, J. Okabayashi, H. Kumigashira, M. Oshima, Y. Sugita, G. L. Liu, Z. Liu, and K. Usuda, Appl. Phys. Lett. **87**, 012903 (2005).
- ¹⁸S. Toyoda, J. Okabayashi, H. Kumigashira, M. Oshima, K. Yamashita, M. Niwa, K. Usuda, and G. L. Liu, J. Appl. Phys. **97**, 104507 (2005).
- ¹⁹E. E. Hoppe, R. S. Sorbello, and C. R. Aita, J. Appl. Phys. **101**, 123534 (2007).
- ²⁰D. Triyoso, R. Liu, D. Roan, M. Ramon, N. V. Edwards, R. Gregory, D. Werho, J. Kulik, G. Tam, E. Irwin, X.-D. Wang, L. B. La, C. Hobbs, R. Garcia, J. Baker, B. E. White, Jr., and P. Tobin, J. Electrochem. Soc. **151**, F220 (2004).
- ²¹D.-Y. Cho, J.-M. Lee, S.-J. Oh, H. Jang, J.-Y. Kim, J.-H. Park, and A. Tanaka, Phys. Rev. B 76, 165411 (2007).