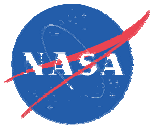


# Partial Thermodynamic Properties of $\gamma'$ -(Ni,Pt)<sub>3</sub>Al

**Evan Copland**

**NASA Glenn / Case Western Reserve University,  
Cleveland Ohio**

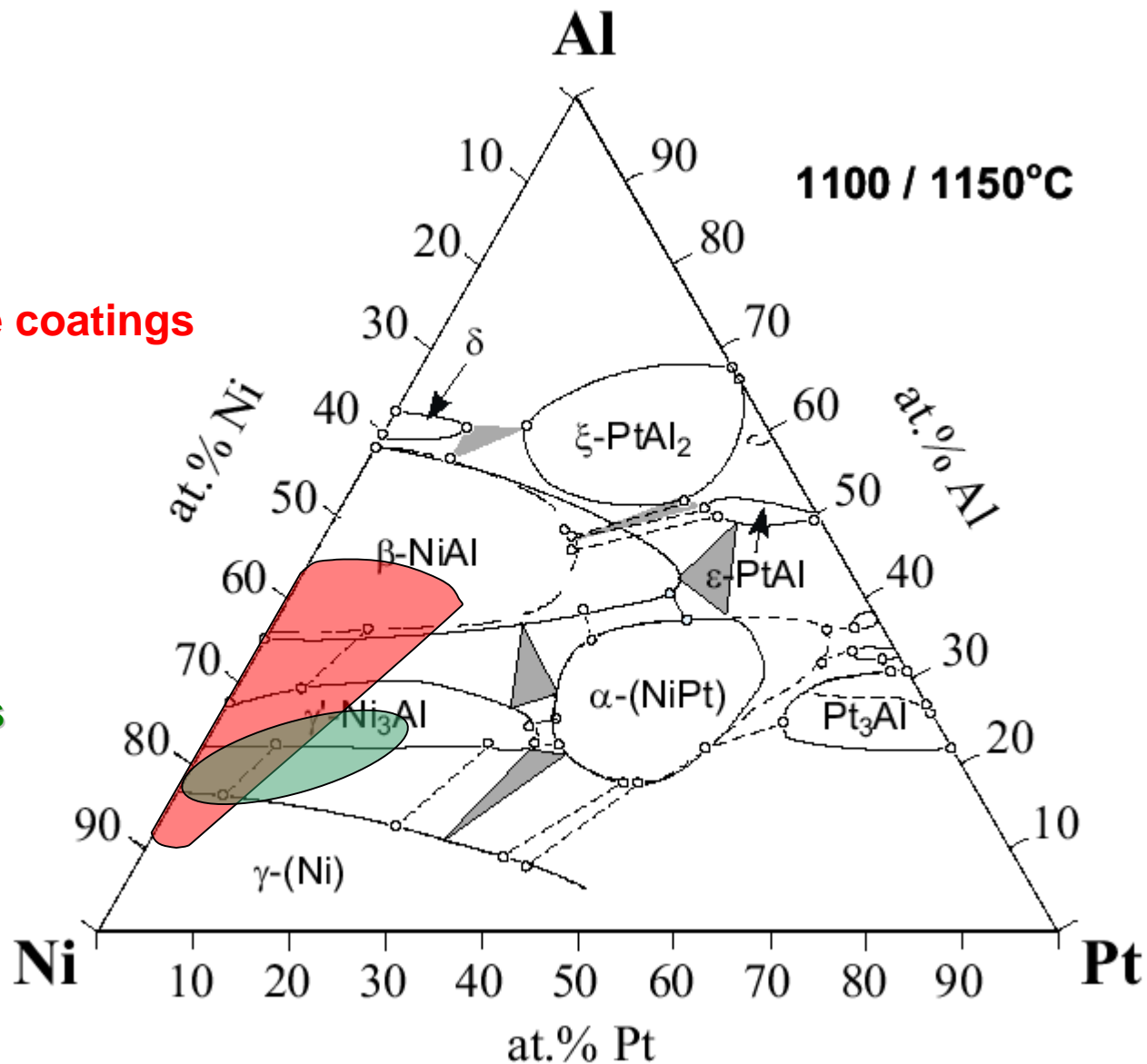
# outline

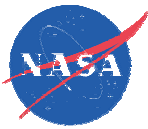


- Ni-Al-Pt phase diagram / scope of study
- partial thermodynamic property measurements
  - ↳ “vapor pressure” technique, multi-cell *KEMS*
  - ↳  $\text{Al}_2\text{O}_3$  container:  $\text{Ni-Al-Pt} \Leftrightarrow \text{Ni-Al-Pt-O}$
  - ↳ reference states:  $\{ \text{Al(l)} + \text{Al}_2\text{O}_3 \}$  and  $\{ \text{Ni(s)} + \text{Al}_2\text{O}_3 \}$
- $a(\text{Al})$  and  $a(\text{Ni})$  in  $\gamma'-(\text{Ni,Pt})_3\text{Al}$ 
  - ↳  $X_{\text{Al}} = 0.24$  (*hypo-stoich.*);  $X_{\text{Pt}} = 0.02 - 0.25$
  - ↳  $X_{\text{Al}} = 0.27$  (*hyper-stoich.*);  $X_{\text{Pt}} = 0.02 - 0.25$
- summarize influence of Pt on Al and Ni
- diffusion profile question...

# Ni-Al-Pt

1100 / 1150°C



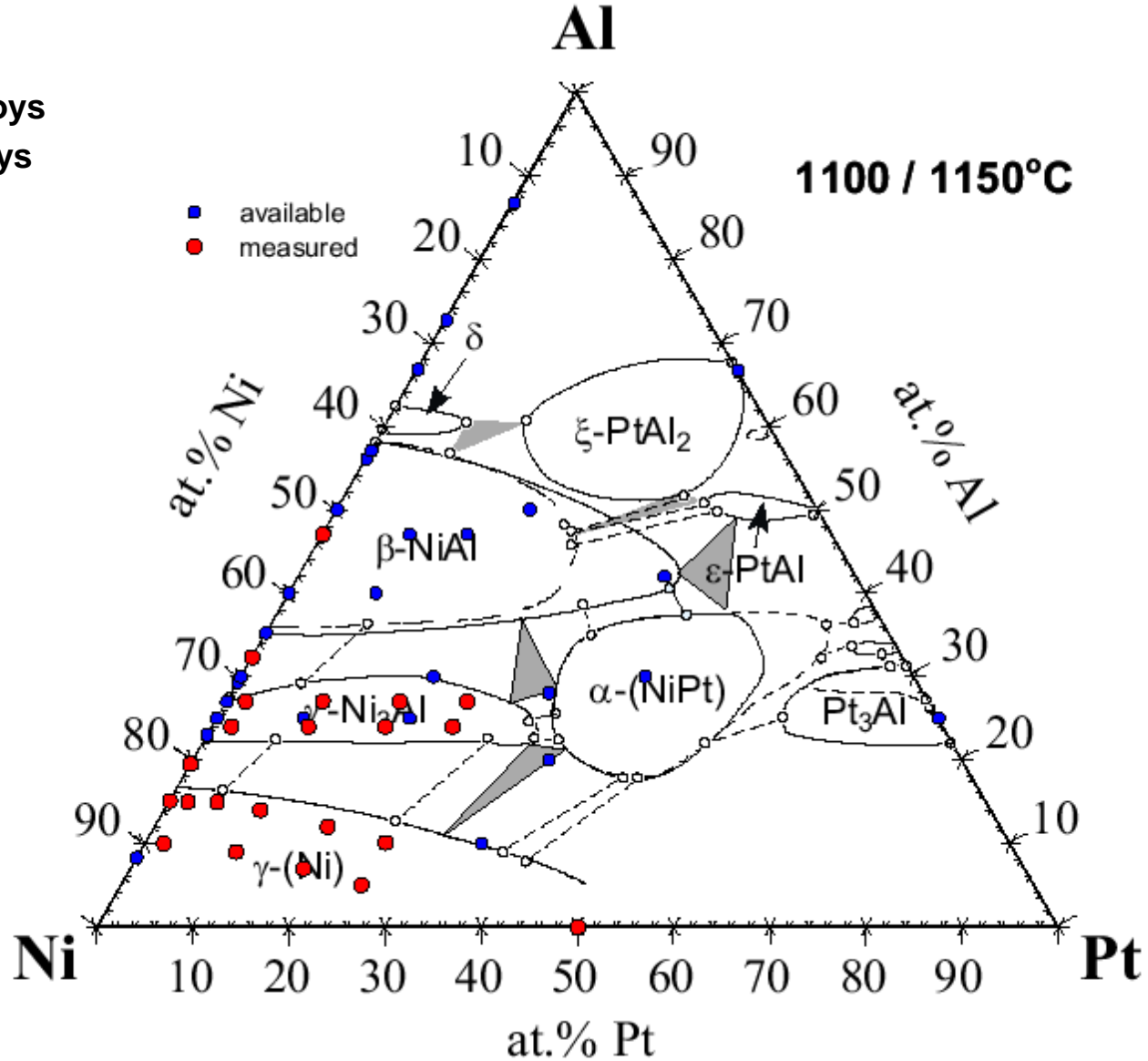


# scope of work

- investigate “Pt effect”... multi-component diffusion
- accurate thermodynamic description of Ni-Al-Pt-X
  - ↳ routine thermodynamic properties measurements
  - ↳ integrate experimental and modeling approach
- initial focus: Ni-Al-Pt... (including Ni-Pt, Al-Pt and Ni-Al)
- future: addition of Cr, Hf, Y,...

# alloy compositions

~30 ternary alloys  
~20 binary alloys

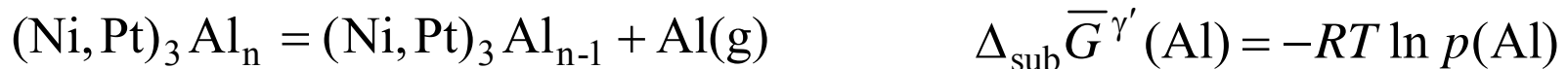




# “vapor pressure technique”

comparing the sublimation behavior...

**solution-phase:  $\gamma'$ -(Ni,Pt)<sub>3</sub>Al**



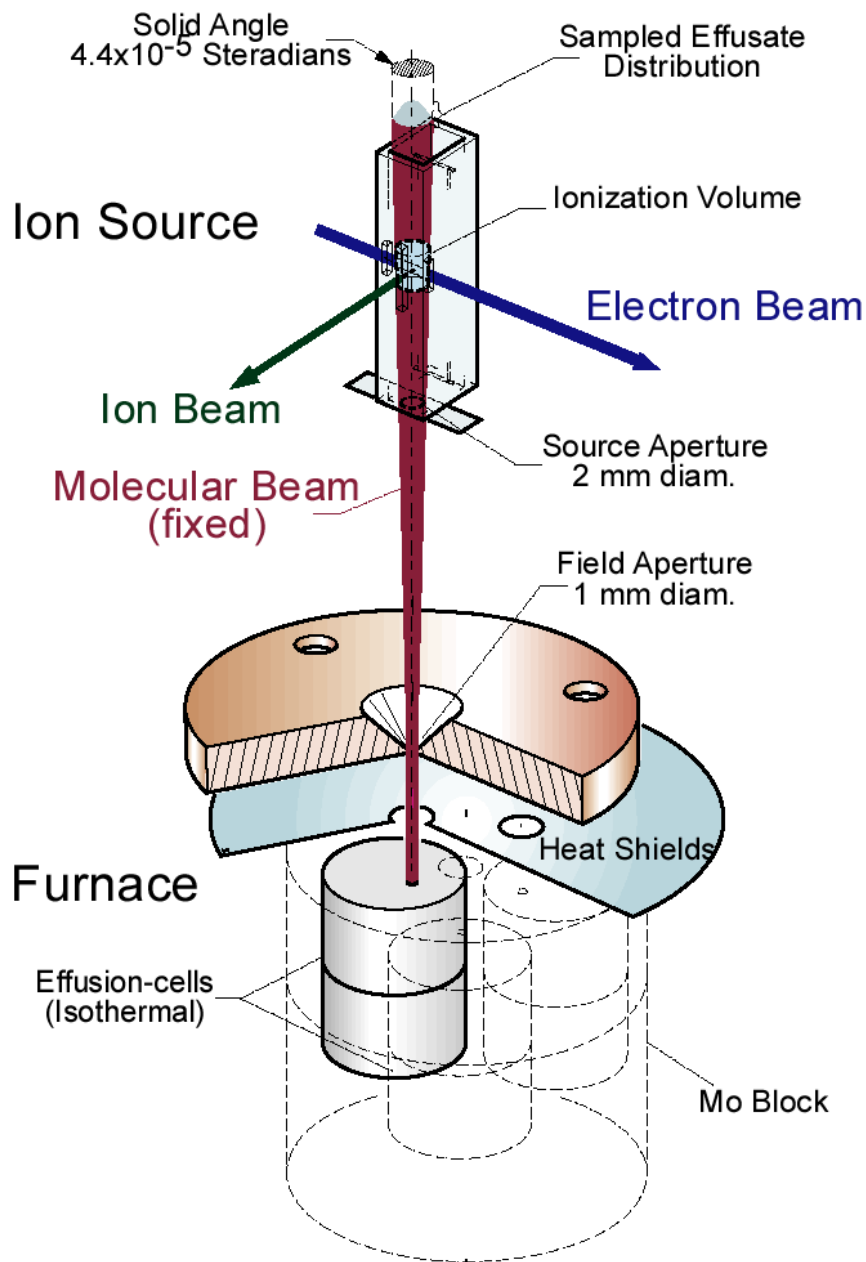
**reference: Al(l)**



$$a(\text{Al}) = \frac{p(\text{Al})}{p^\circ(\text{Al})}$$

**measure pressure ratio as function ( *comp.*, *T* )**

# multi-cell *KEMS*



**absolute pressures**

$$p(i) = I_{ik}^+ T / S_{ik}$$

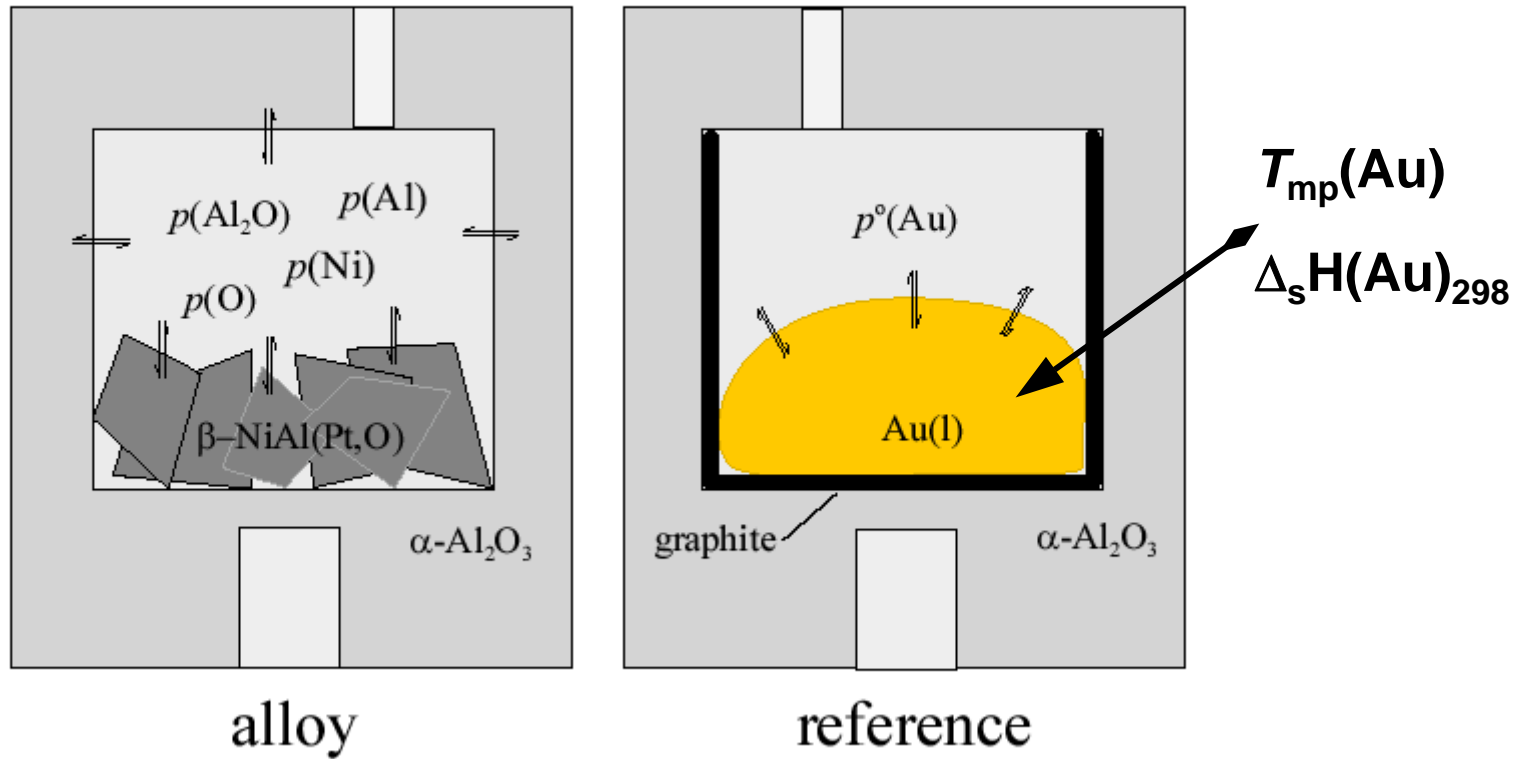
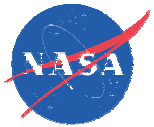
**sensitivity coefficient**

$$S_{ik} = g \varepsilon_k(x, y, z) \tau_k \sigma_{ik}(E) \gamma_k f_i$$

**relative pressures**

$$p(i) \propto I_{ik}^+ T$$

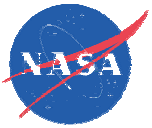
# indirect measurement



$$a(i) = \frac{p(i)}{\cancel{p^\circ(\text{Au})}} \cdot \left[ \frac{\cancel{p^\circ(\text{Au})}}{p^\circ(i)} \right] = \frac{I_i}{I_{\text{Au}}^\circ} \cdot \left[ \frac{S_{\text{Au}}}{S_i} \cdot \frac{g(R)}{g(A)} \cdot \left[ \frac{p^\circ(\text{Au})}{p^\circ(i)} \right] \right] \quad (i = \text{Ni}, \text{Al}, \text{Al}_2\text{O})$$



# reference state calibrations...



$$Au(s,l) = Au(g) \quad K_1 = p^\circ(Au)$$

$$Al(l) = Al(g) \quad K_2 = p^\circ(Al)$$

$$4/3 Al(l) + 1/3 Al_2O_3(s) = Al_2O(g) \quad K_3 = p^\circ(Al_2O)$$



$$Au(s,l) = Au(g) \quad K_1 = p^\circ(Au)$$

$$Ni(s) = Ni(g) \quad K_4 = p^\circ(Ni)$$

$$\frac{p^\circ(i)}{p^\circ(Au)} = \boxed{\left[ \frac{p^\circ(i)}{p^\circ(Au)} \right]} \quad \boxed{\frac{S_{Au}}{S_i}} \left( \text{or } \frac{\sigma_{Au}(E)\gamma_{Au}}{\sigma_i(E)\gamma_i} \right) = \frac{I^\circ_{Au}}{I_i^\circ} \cdot \frac{g(i)}{g(Au)} \cdot \boxed{\left[ \frac{p^\circ(i)}{p^\circ(Au)} \right]}$$

# improved { Al(l) + Al<sub>2</sub>O<sub>3</sub> } data

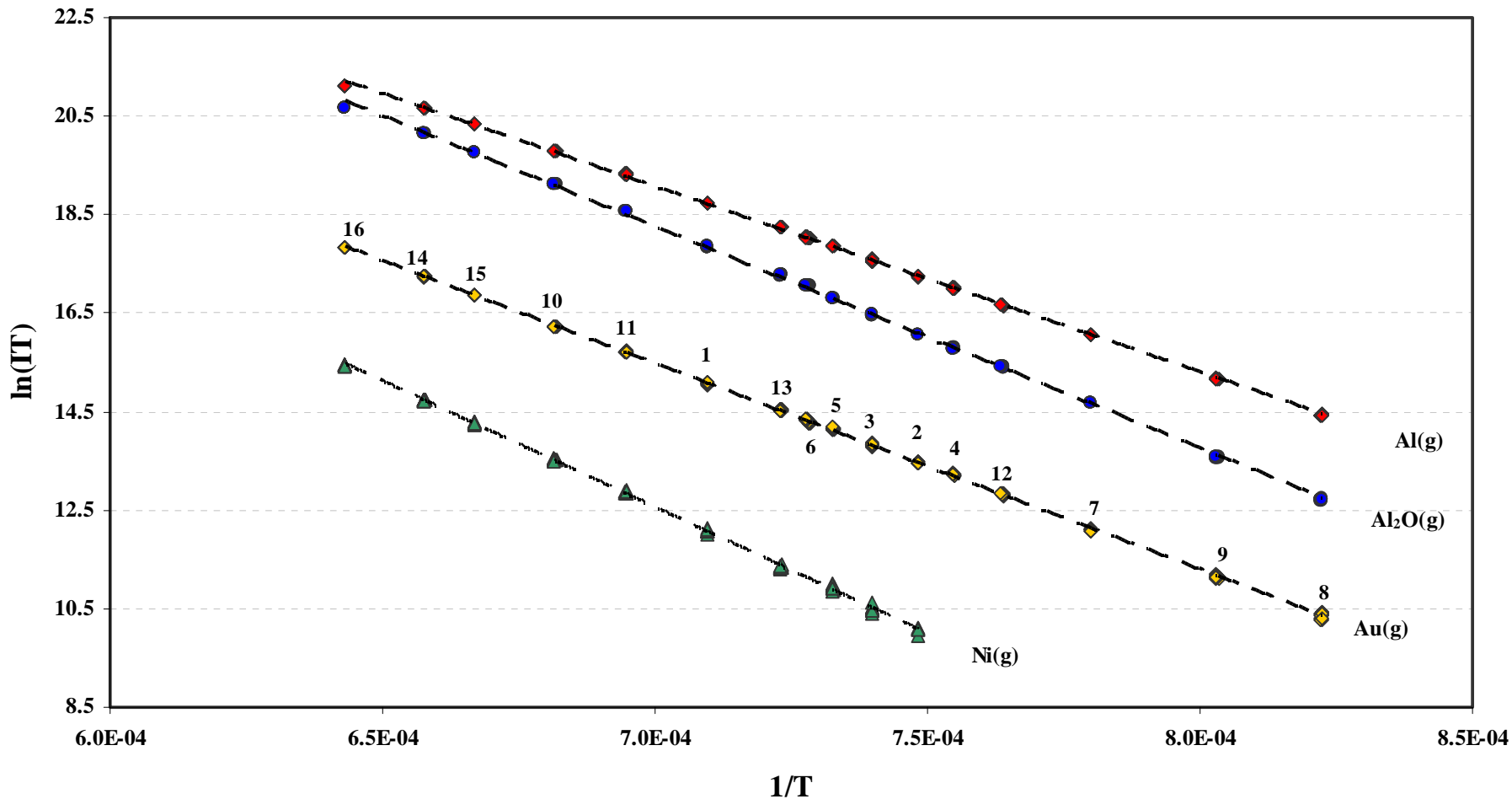


reaction (298.15K)	measured (kJmol <sup>-1</sup> )	IVTAN (kJmol <sup>-1</sup> )	JANAF (kJmol <sup>-1</sup> )
Au(s,l) = Au(g)	363.5±2.8 367.0±1.3*	367.0±0.9	
Ni(s) = Ni(g)	428.3±2.6	428.0±8.0	430.1±8.4
Al(s) = Al(g)	341.0±2.2	330.0±3.0	329.7±4.2
4/3Al(s) + 1/3Al <sub>2</sub> O <sub>3</sub> (s) = Al <sub>2</sub> O(g)	414.2±3.6	409.9±55	413.4±50
4/3Al(g) + 1/3Al <sub>2</sub> O <sub>3</sub> (s) = Al <sub>2</sub> O(g)	-41.1±3.2	-30.0±4.3	-26.2±3.0
2Al(g) + O(g) = Al <sub>2</sub> O(g)	-1075.5±9.0	-1057.8±20.0	-1053.7±150

\* 3<sup>rd</sup> law measurements

- ignore tabulated data and use measured 2<sup>nd</sup> law data...
- $p^\circ(\text{Al})$  and  $p^\circ(\text{Al}_2\text{O})$  for { Al(l) + Al<sub>2</sub>O<sub>3</sub> } not well known
- Au(s,l) ref. → allows measurement of 2 alloys in each exp.

# reference state calibrations...

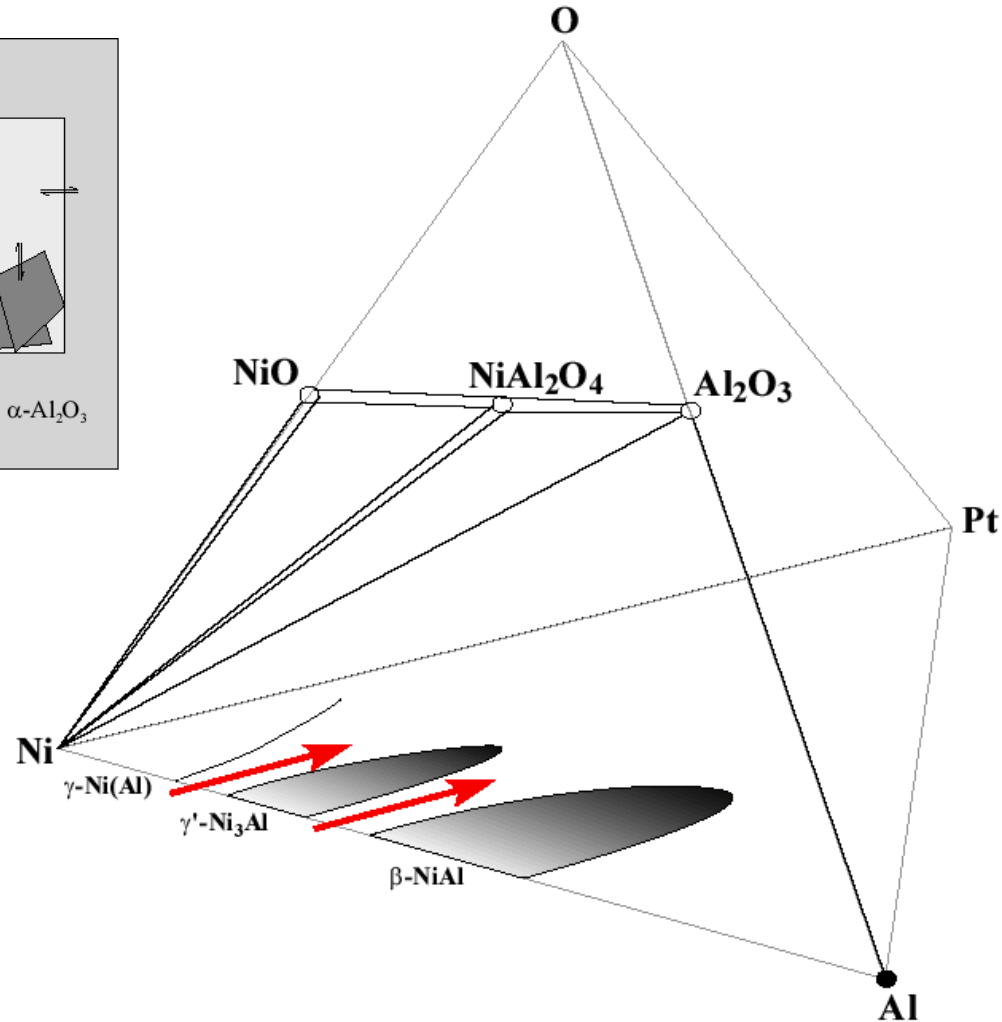
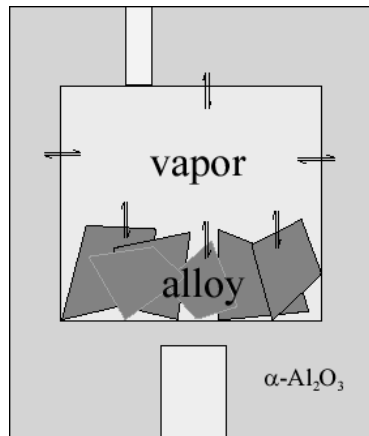


$$\frac{S_{Au}}{S_{Al}} = 0.154 \pm 0.003$$

$$\frac{S_{Au}}{S_{Al_2O}} = 0.506 \pm 0.02$$

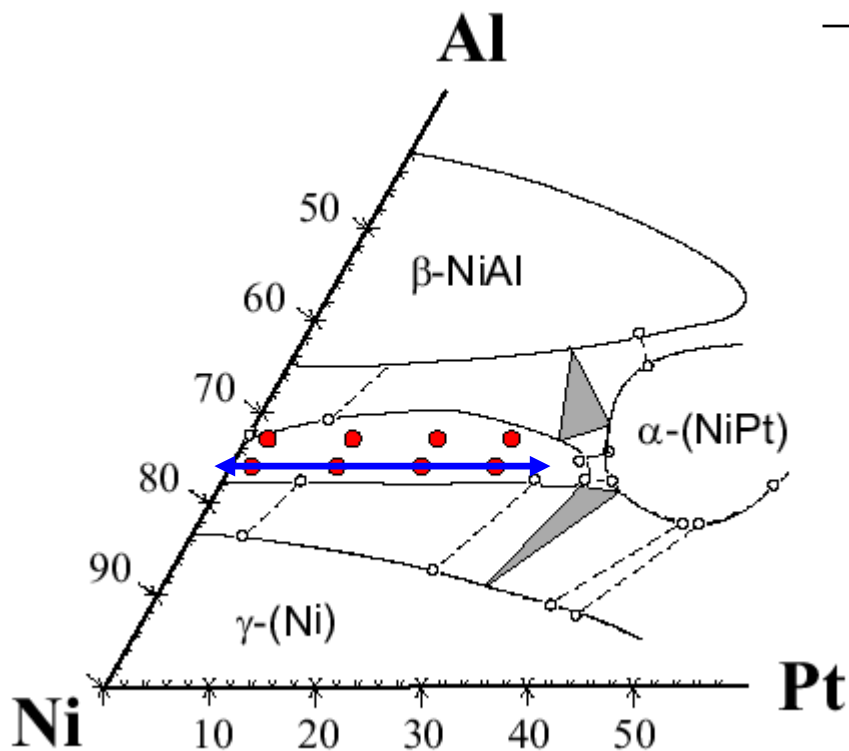
$$\frac{S_{Au}}{S_{Ni}} = 0.85 \pm 0.03$$

# Ni-Al-Pt-O $\Leftrightarrow$ Ni-Al-Pt



- actually studying { alloy +  $\text{Al}_2\text{O}_3$  } equilibrium...
- limited O solubility in alloy; limited  $\text{Al}_2\text{O}_3$  stoichiometry

# $\gamma'$ -(Ni,Pt)<sub>3</sub>Al, $X_{Al} = 0.24$

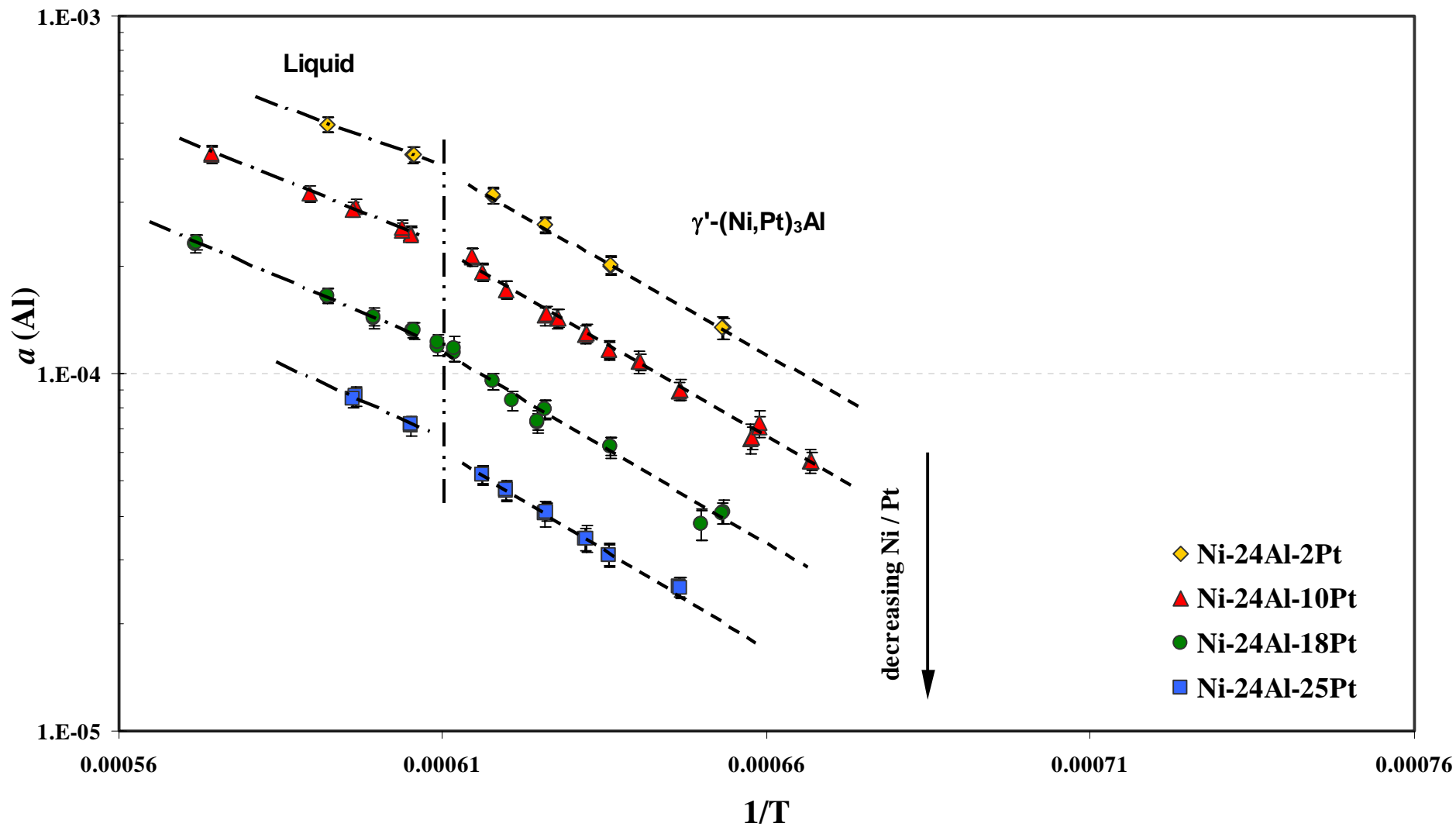


Ni	Al	Pt	Ni/Pt	Al/Pt
73.6	24.3	2.0	35.05	11.57
65.8	24.2	10.0	6.58	2.42
57.9	24.0	18.1	3.2	1.33
51.1	23.8	25.1	2.04	0.95

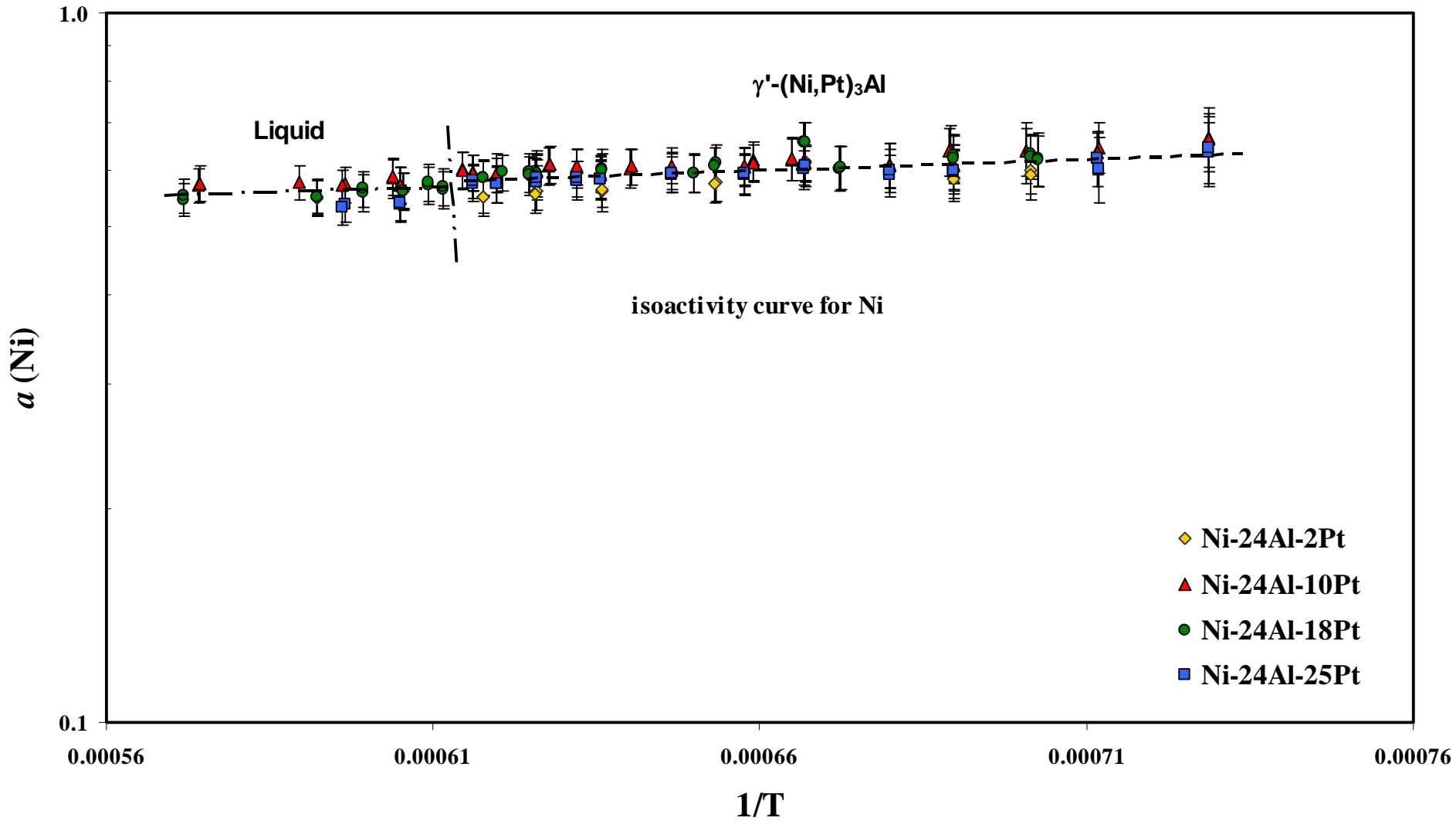
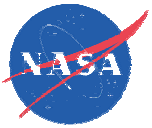
( at.%  $\pm$  0.5 )



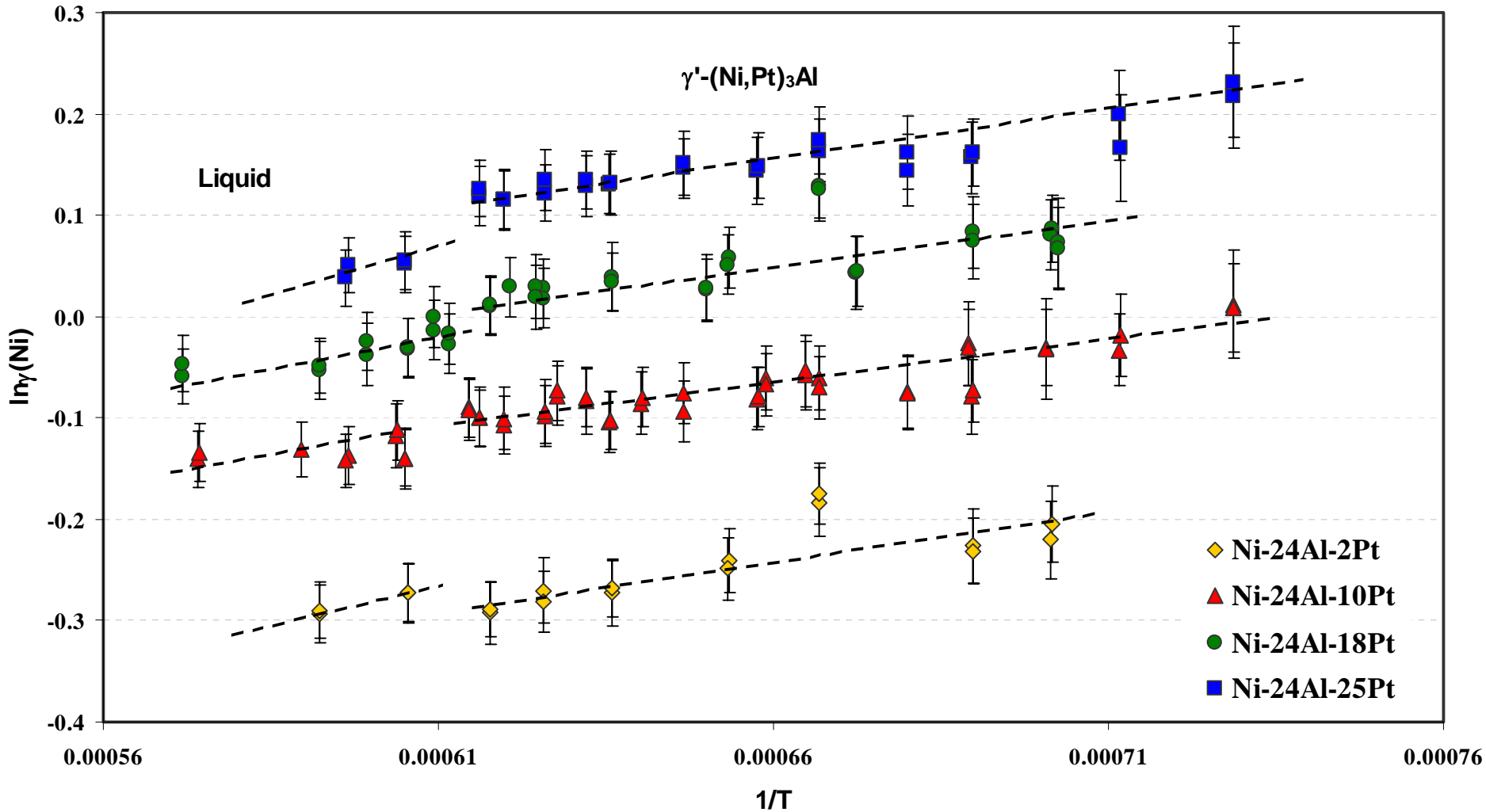
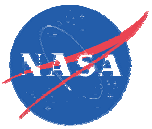
# $a(\text{Al})$ vs $1/T$ (Ni-24Al-XPt)



# $a(\text{Ni})$ vs $1/T$ (Ni-24Al-XPt)



# $\ln\gamma(\text{Ni})$ vs $1/T$ (Ni-24Al-XPt)







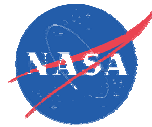
# partial mixing enthalpies

$$\Delta_m \bar{H}(i) = R \frac{\partial \ln a(i)}{\partial 1/T}$$

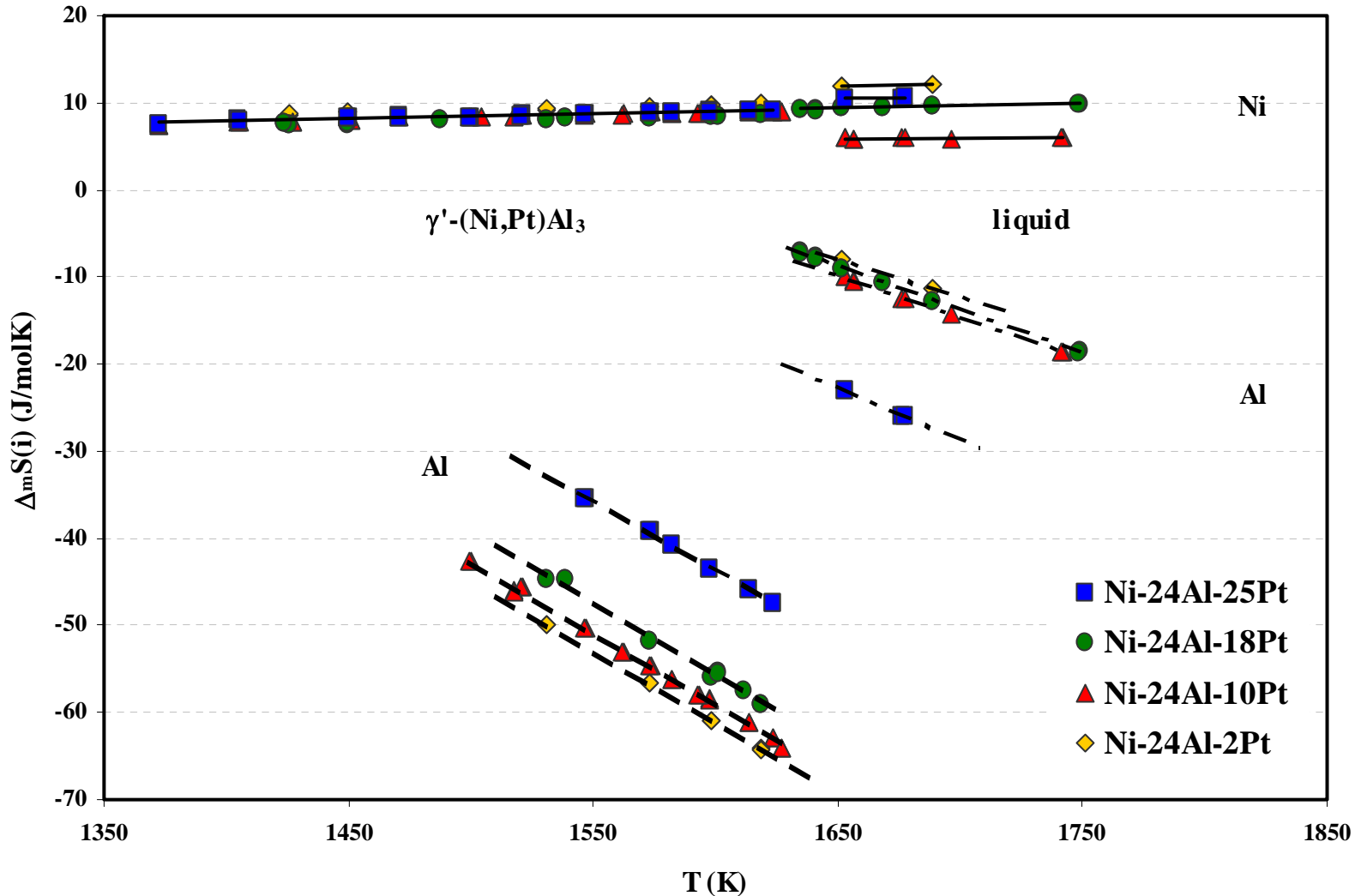
$\gamma'-(\text{Ni,Pt})_3\text{Al}$	$\Delta_m \bar{H}(\text{Ni})$ (kJ/mol)	T (K)	$\Delta_m \bar{H}(\text{Al})$ (kJ/mol)	T (K)	$T_{\text{mp}}$ (K)
Ni-24Al-2Pt	$7.1 \pm 1.3$ <sub>12</sub>	1522	$-200.5 \pm 5.8$ <sub>8</sub>	1575	1619-1651
Ni-24Al-10Pt	$6.7 \pm 1.2$ <sub>38</sub>	1500	$-201.9 \pm 8.0$ <sub>24</sub>	1563	1627-1651
Ni-24Al-18Pt	$6.1 \pm 1.3$ <sub>20</sub>	1521	$-207.8 \pm 25$ <sub>13</sub>	1575	1619-1635
Ni-24Al-25Pt	$6.4 \pm 1.2$ <sub>24</sub>	1498	$-200.2 \pm 11$ <sub>12</sub>	1585	1623-1653

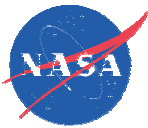
<b>liquid</b>	$\Delta_m \bar{H}(\text{Ni})$ (kJ/mol)	T (K)	$\Delta_m \bar{H}(\text{Al})$ (kJ/mol)	T (K)
Ni-24Al-2Pt	$11.9 \pm 5.0$ <sub>4</sub>	1670	$-123.1 \pm 5.4$ <sub>4</sub>	1670
Ni-24Al-10Pt	$2.3 \pm 6.0$ <sub>9</sub>	1697	$-137.9 \pm 7.8$ <sub>9</sub>	1697
Ni-24Al-18Pt	$8.3 \pm 4.0$ <sub>12</sub>	1692	$-144.3 \pm 5.2$ <sub>12</sub>	1692
Ni-24Al-25Pt	$8.9 \pm 20$ <sub>4</sub>	1665	$-172 \pm 56$ <sub>4</sub>	1665

# partial entropy of mixing: Al, Ni



$$\Delta_m \bar{S}(i) = -R \ln a(i) - RT \frac{\partial \ln a(i)}{\partial T}$$

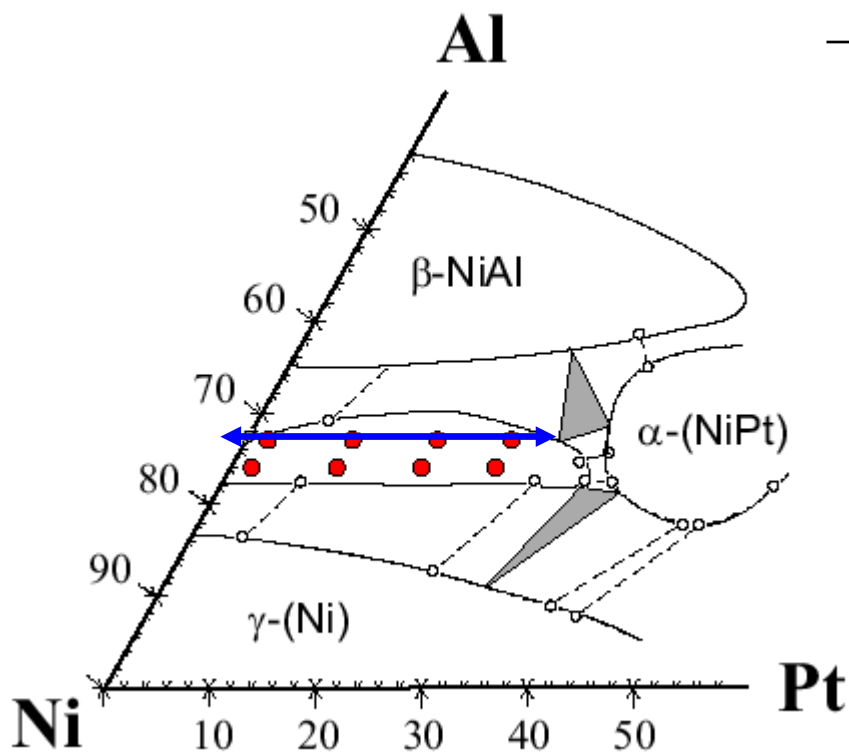




# Ni-24Al-XPt...

- $\gamma'-(\text{Ni,Pt})_3\text{Al}$ :  $a(\text{Al})$  decreases  $\sim 10$  with Pt addition (Ni/Pt  $\downarrow$ )
- $\Delta_m H(\text{Al}) -203 \pm 10 \text{ kJmol}^{-1}$ , independent of Ni/Pt
  - ↳ bond strength:  $\text{Al} \leftrightarrow \text{Ni}$  identical to  $\text{Al} \leftrightarrow \text{Pt}$  (?)
  - ↳ or  $X_{\text{Al}}$  has strongest influence on Al bonding
- $\Delta_m S(\text{Al}) -60$  to  $-40 \text{ Jmol}^{-1}\text{K}^{-1}$  with Pt addition (Ni/Pt  $\downarrow$ )
  - ↳ increase in  $S(\text{Al}) \rightarrow$  decreasing  $a(\text{Al})$
  - ↳ entropy-based interaction between Ni- and Al-lattices (?)
- liquid:  $a(\text{Al})$  decreases with Pt addition (Ni/Pt  $\downarrow$ )
- $\Delta_m H(\text{Al}) -120$  to  $-170 \text{ kJmol}^{-1}$ ,  $\Delta_m S(\text{Al}) -10$  to  $-25 \text{ Jmol}^{-1}\text{K}^{-1}$ 
  - ↳ increasing Al ordering in liquid with Pt addition
- $a(\text{Ni})$  constant with Pt addition (Ni/Pt  $\downarrow$ ):  $\gamma_{\text{Ni}}$  increases 0.7 to 1.2
  - ↳  $\Delta_m H(\text{Ni}) \sim 7 \text{ kJmol}^{-1}$ ,  $\Delta_m S(\text{Al})$  +ve
  - ↳ +ve ternary interaction between Ni  $\leftrightarrow$  (Al + Pt)

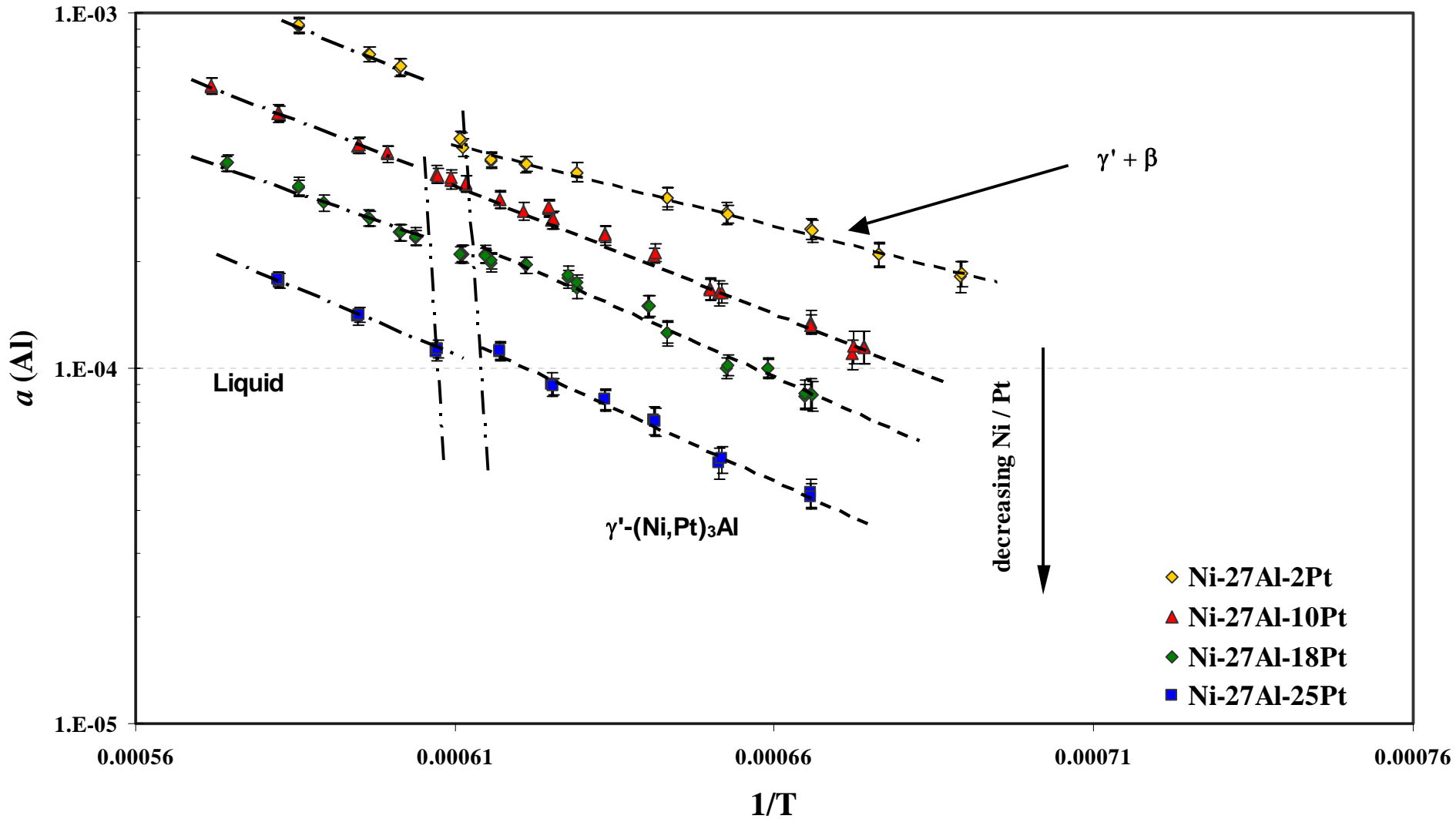
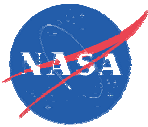
# $\gamma'$ -(Ni,Pt)<sub>3</sub>Al, $X_{Al} = 0.27$



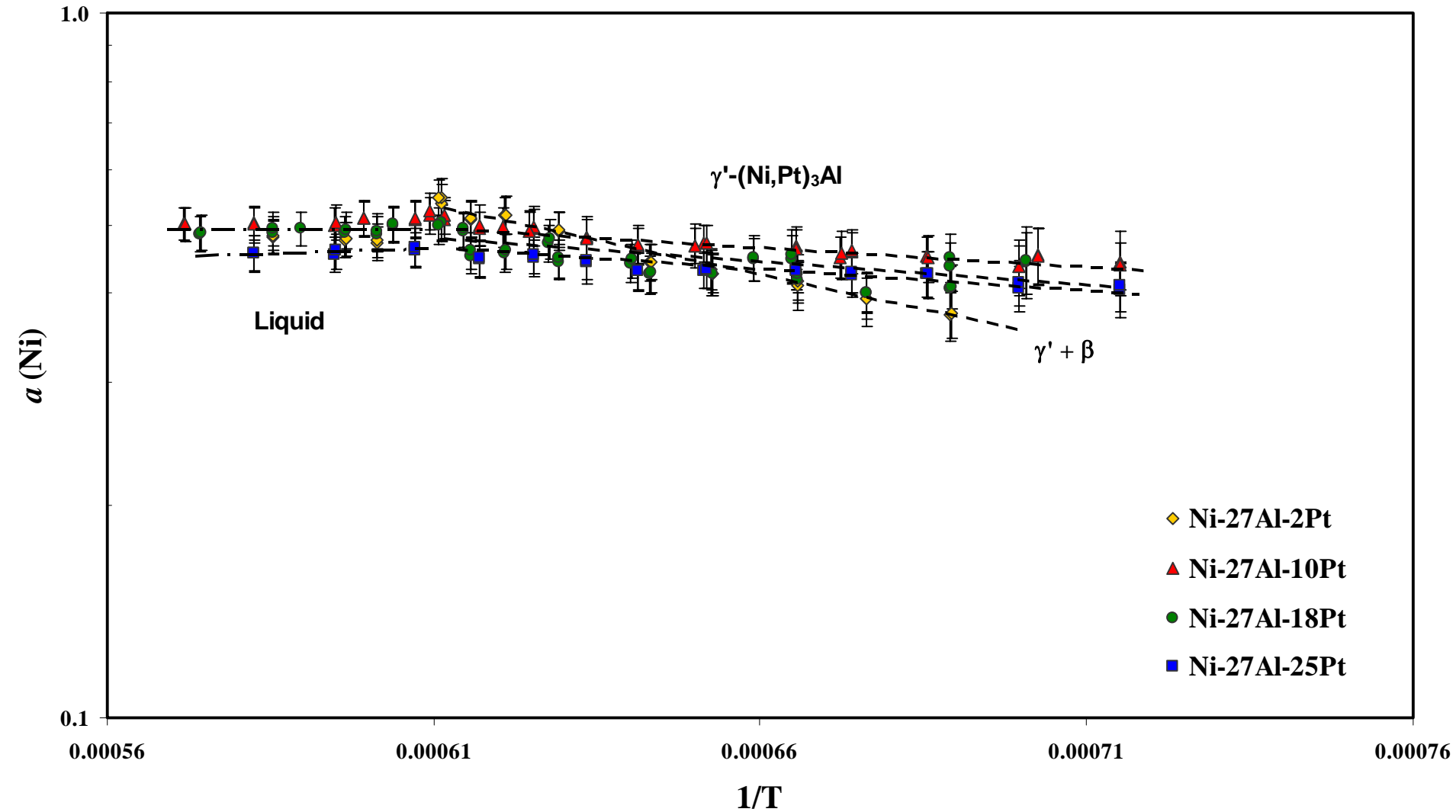
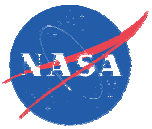
Ni	Al	Pt	Ni/Pt	Al/Pt
70.8	27.2	2.0	35.4	13.6
63.8	26.4	9.8	6.51	2.69
54.9	27.0	18.1	3.03	1.49
48.1	26.7	25.2	1.91	1.06

( at.%  $\pm$  0.5 )

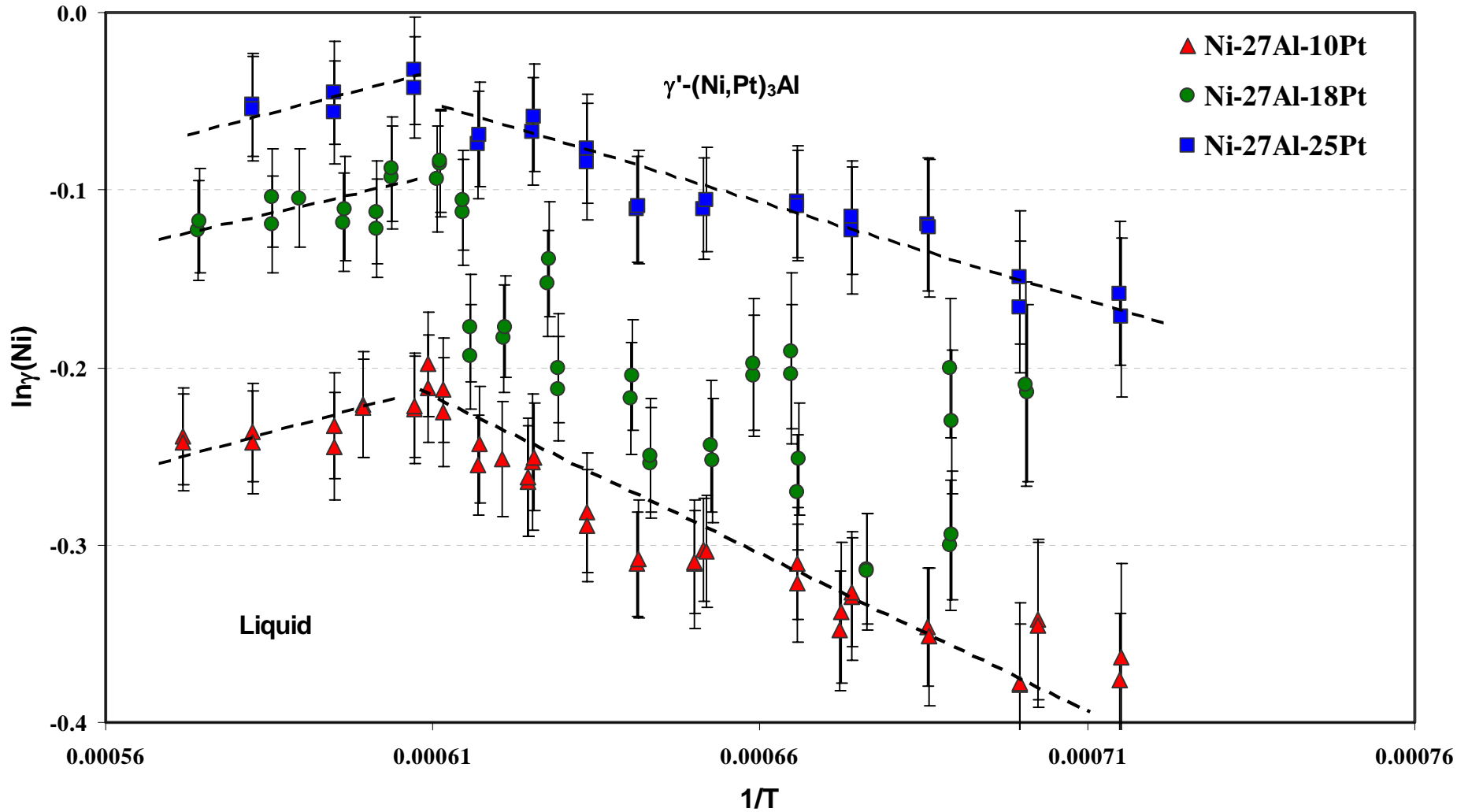
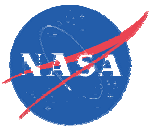
# $a(\text{Al})$ vs $1/T$ (Ni-27Al-XPt)



# $a(\text{Ni})$ vs $1/T$ (Ni-27Al-XPt)



# $\ln\gamma(\text{Ni})$ vs $1/T$ (Ni-27Al-XPt)





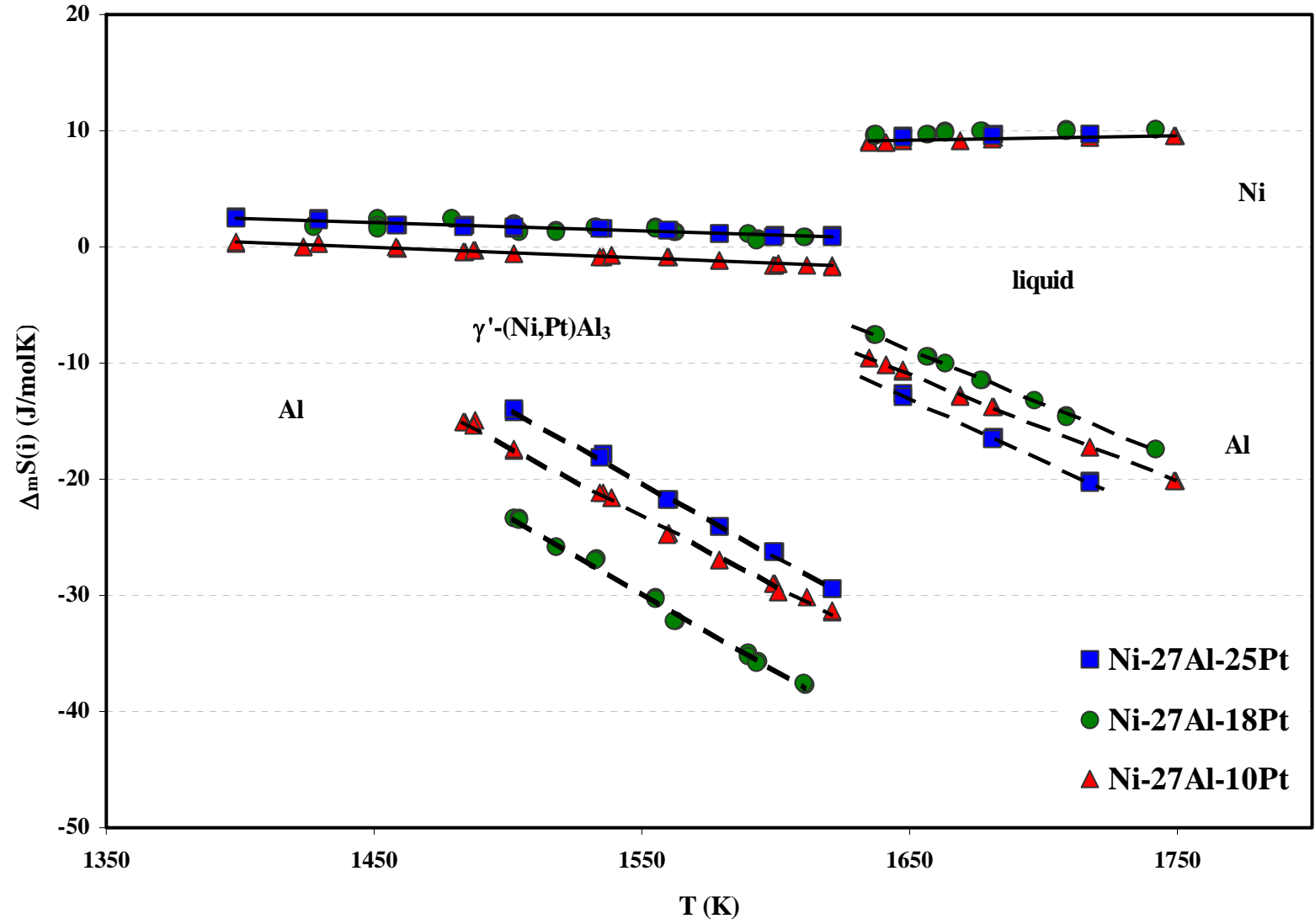
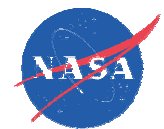
# partial mixing enthalpies

$\gamma'-(\text{Ni,Pt})_3\text{Al}$	$\Delta_m \bar{H}(\text{Ni})$ (kJ/mol)	T (K)	$\Delta_m \bar{H}(\text{Al})$ (kJ/mol)	T (K)	$T_{\text{mp}}$ (K)
Ni-27Al-2Pt	$-39.5 \pm 2.9$ <sub>19</sub>	1544	$-88.2 \pm 3.9$ <sub>18</sub>	1544	1637
Ni-27Al-10Pt	$-10.5 \pm 1.4$ <sub>29</sub>	1510	$-146.5 \pm 6.8$ <sub>21</sub>	1552	1620-1634
Ni-27Al-18Pt	$-7.8 \pm 5.5$ <sub>25</sub>	1519	$-162.7 \pm 12.1$ <sub>18</sub>	1556	1610-1634
Ni-27Al-25Pt	$-8.1 \pm 1.4$ <sub>20</sub>	1509	$-157.9 \pm 11.0$ <sub>12</sub>	1562	1620-1646

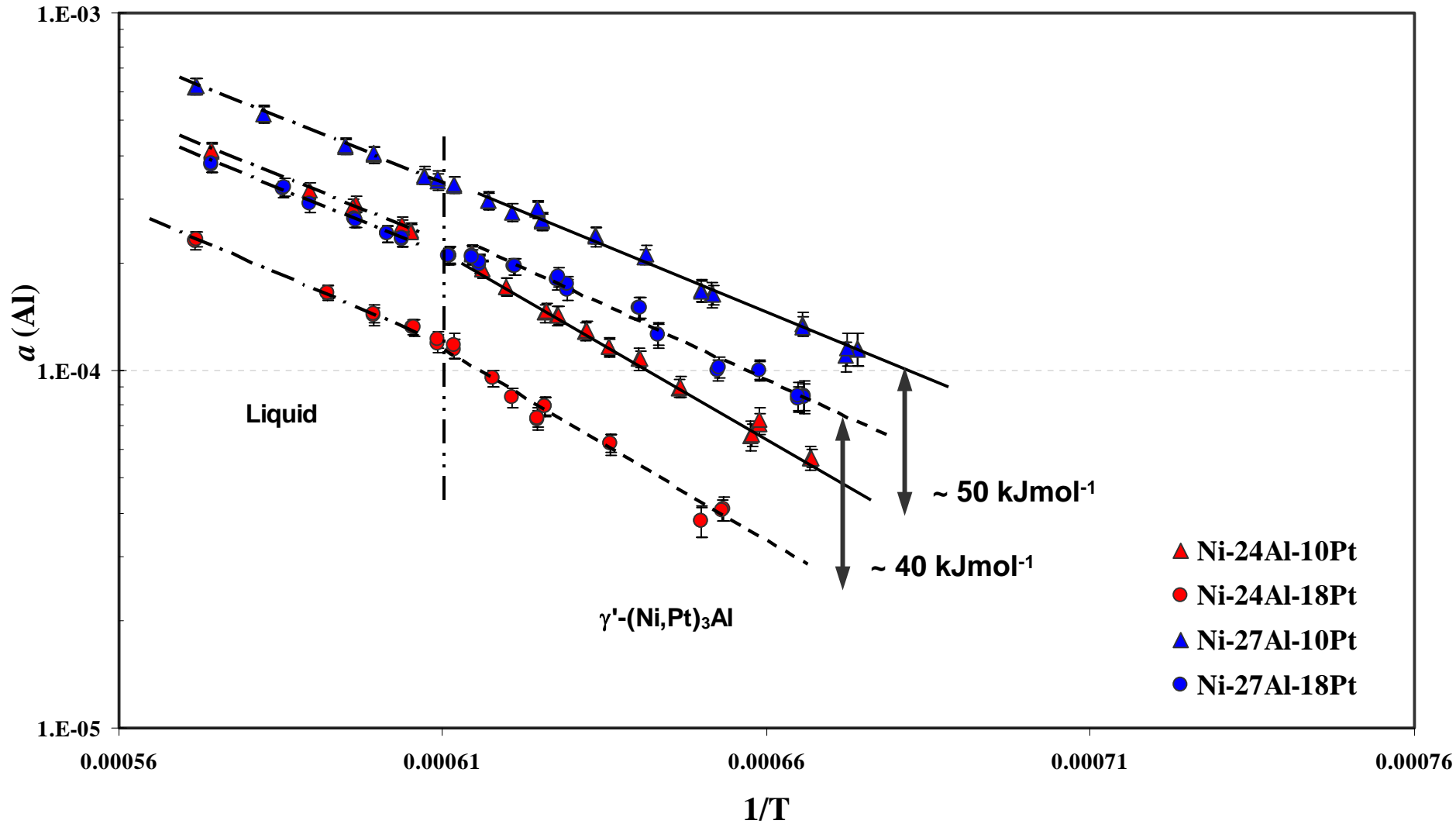
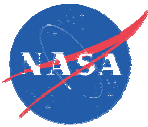
<b>liquid</b>	$\Delta_m \bar{H}(\text{Ni})$ (kJ/mol)	T (K)	$\Delta_m \bar{H}(\text{Al})$ (kJ/mol)	T (K)
Ni-27Al-2Pt	$-8.4 \pm 5.1$ <sub>6</sub>	1686	$-141.0 \pm 3.9$ <sub>6</sub>	1686
Ni-27Al-10Pt	$6.2 \pm 3.1$ <sub>14</sub>	1692	$-133.1 \pm 3.0$ <sub>14</sub>	1692
Ni-27Al-18Pt	$6.9 \pm 4.3$ <sub>14</sub>	1689	$-135.6 \pm 4.9$ <sub>14</sub>	1689
Ni-27Al-25Pt	$5.3 \pm 5.7$ <sub>6</sub>	1682	$-151.7 \pm 9.7$ <sub>6</sub>	1682



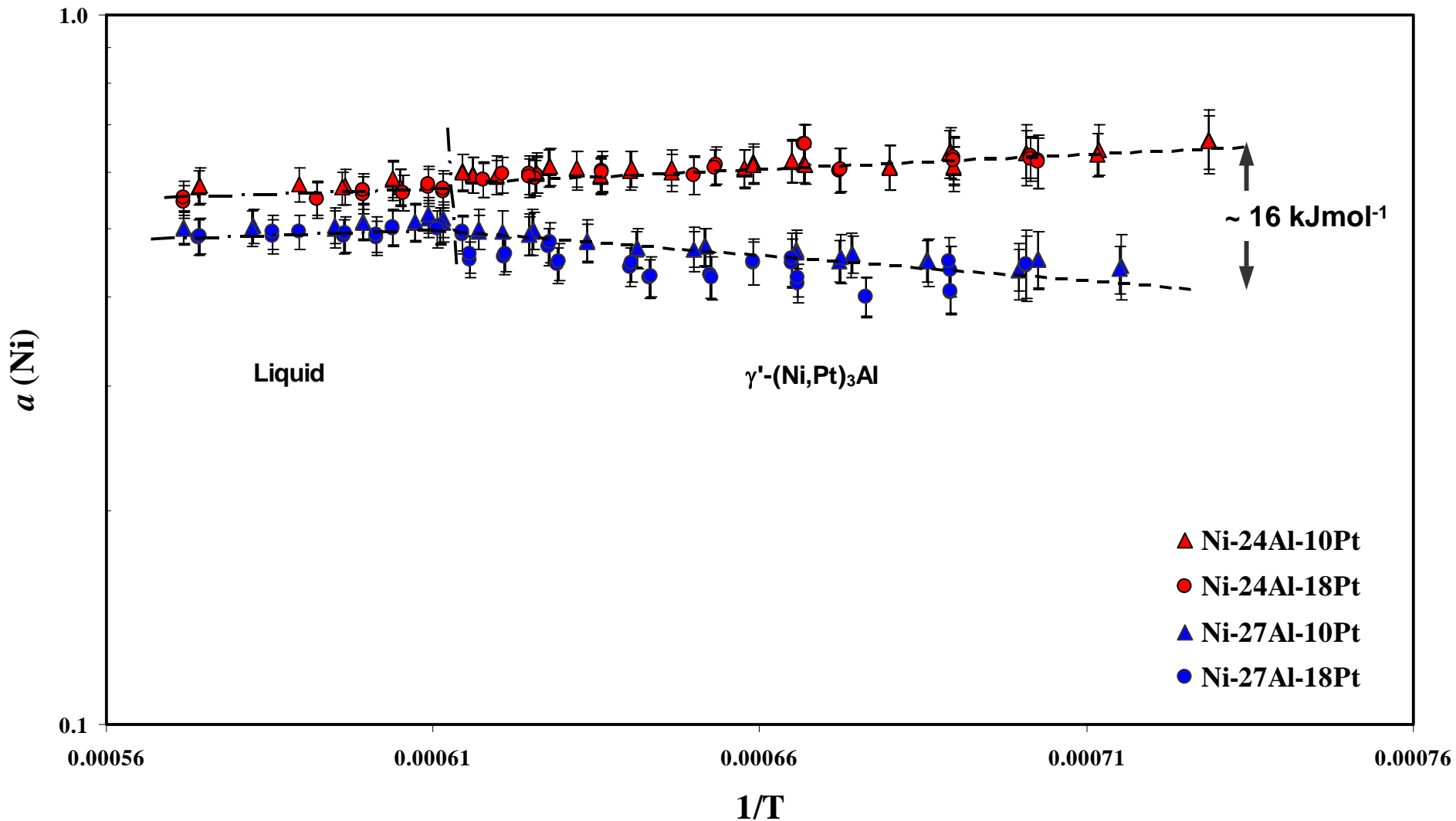
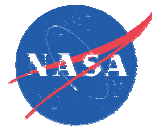
# partial entropy of mixing: Al, Ni

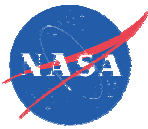


# hypo- / hyper-stoichiometric: $a(\text{Al})$



# hypo- / hyper-stoichiometric: $a(\text{Ni})$

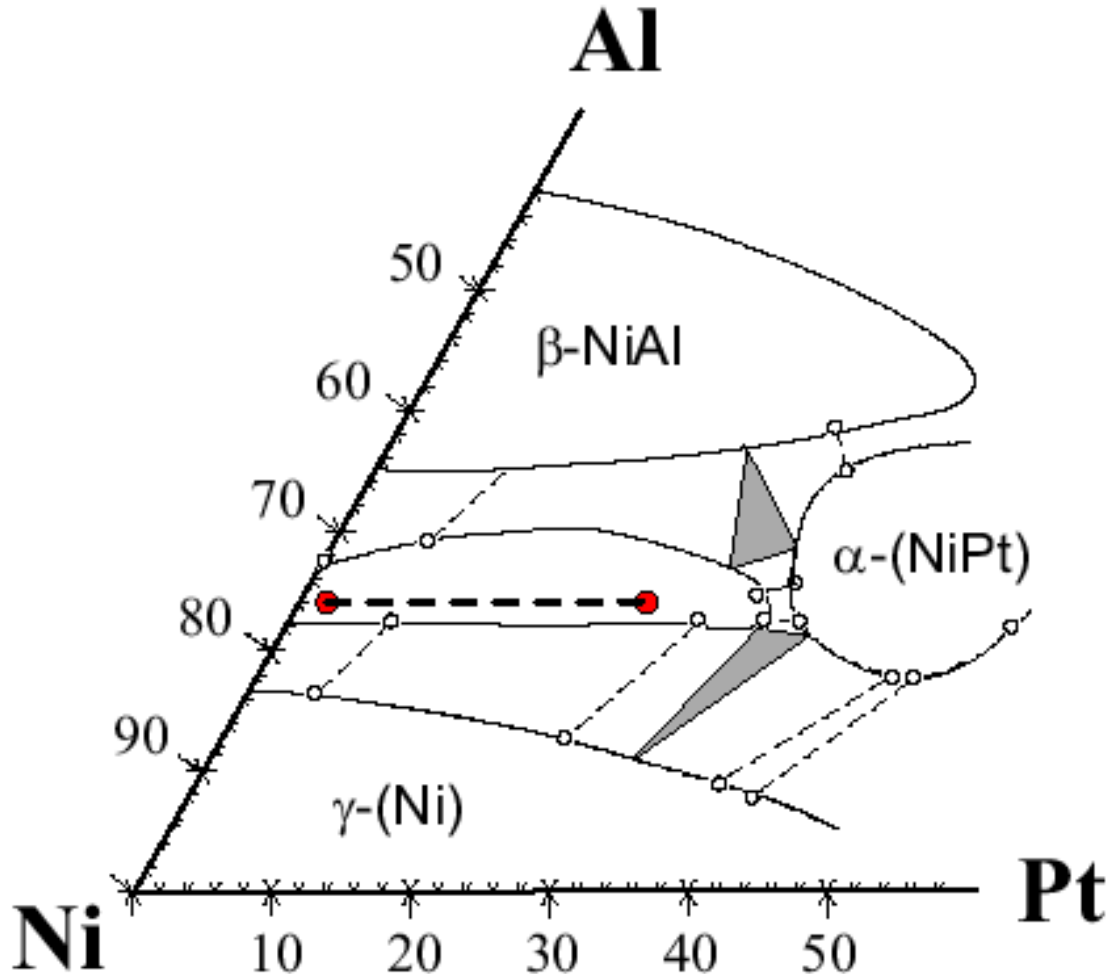
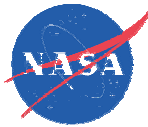




# Ni-27Al-XPt, summary

- $\gamma'-(Ni,Pt)<sub>3</sub>Al:  $a(\text{Al})$  decreases  $\sim 3$  with Pt addition (Ni/Pt  $\downarrow$ )$
- $X_{\text{Al}}$  strongest influence on Al bonding
- $a(\text{Al})$  relation with  $S(\text{Al})$  not so clear
  - ↳ decrease ordering of Al-atoms
- $a(\text{Ni})$  constant with Pt (Ni/Pt  $\downarrow$ ):  $\gamma_{\text{Ni}}$  increased from 0.7 to 0.95
  - ↳  $\Delta_{\text{m}}H(\text{Ni}) -9 \pm 3.0 \text{ kJmol}^{-1}$ ,  $\Delta_{\text{m}}S(\text{Ni})$  -ve
  - ↳ increased ordering of Ni-atoms
  - ↳ +ve ternary interaction between Ni  $\leftrightarrow$  (Al + Pt)
- liquid behavior remains similar
- dramatic change in mixing behavior for:  $X_{\text{Al}} = 0.24$  and  $0.27$ 
  - ↳  $\Delta_{\text{m}}H(\text{Al}) \uparrow 50 \text{ kJmol}^{-1}$ ,  $\Delta_{\text{m}}H(\text{Ni}) \downarrow 16 \text{ kJmol}^{-1}$
  - ↳  $\Delta_{\text{m}}S(\text{Al}) \uparrow 20 \text{ Jmol}^{-1}\text{K}^{-1}$ ,  $\Delta_{\text{m}}S(\text{Ni})$  +ve to -ve

# interesting diffusion couple...



- $a(\text{Ni}) \sim \text{constant}$
- Al driving force:  
 $S(\text{Al}) \Leftrightarrow \text{Ni-lattice}$
- Pt behavior ?

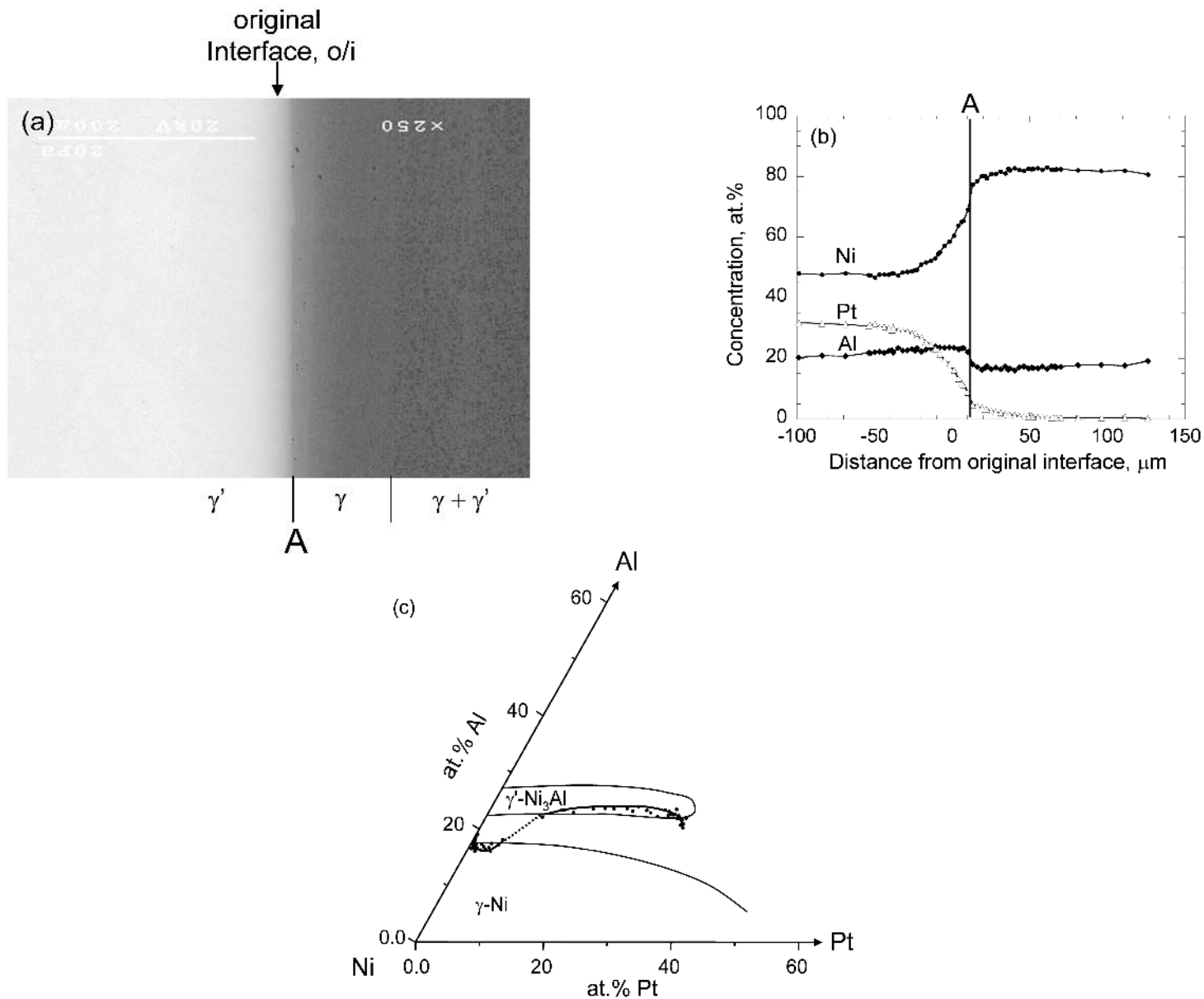
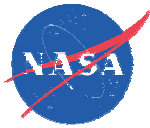


Fig. 2—Cross-sectional and concentration profiles of Ni-22Al-30Pt/Ni-19Al diffusion couple annealed for 10 h at 1150 °C: (a) cross section, (b) concentration profiles of each element, and (c) diffusion path.

# co-workers / acknowledgements



Helpful discussion:

Brian Gleeson (ISU),

Nathan Jacobson and James Nesbitt (NASA Glenn),

Christian Chatillon (Saint Martin d'Hères, France)

David Young (UNSW)

Claude Lupis (MIT)

Pat Martin and Dallis Hardwick (AFRL, Wright-Patterson AFB)

Support for this work from:

**NASA Glenn, Low Emission Alternative Power Project**

**AFRL / MLLM, Materials for Air Breathing Propulsion Project**