

Introduction of Dictra

Zi-Kui Liu Pennsylvania State University

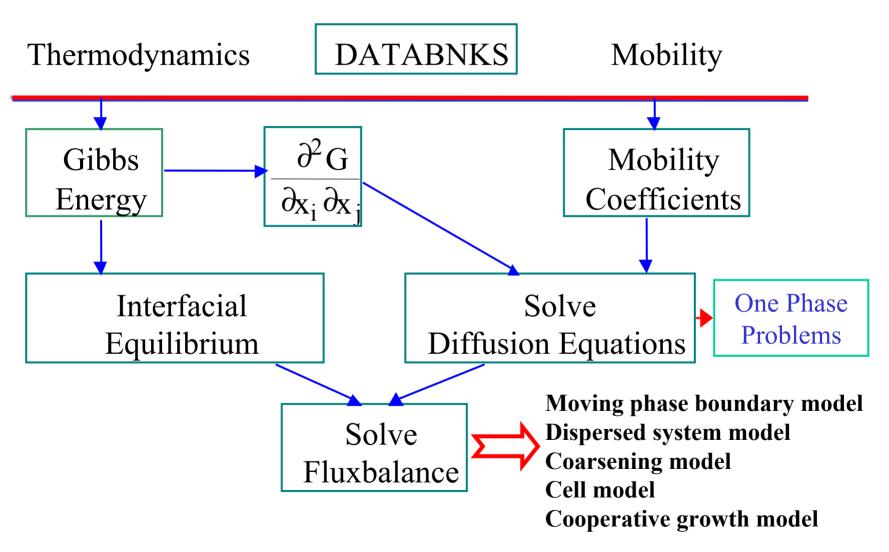
With help from Thermo-Calc Software, Stockholm, Sweden

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What is DICTRA?

- Software package for simulation of diffusional reactions in multicomponent alloys.
- The result of more than 20 years and 60 man-years R&D at KTH in Stockholm, Sweden, in cooperation with the Max-Planck Institute für Eisenforchung in Düsseldorf, Germany.

Numerical Procedure of Dictra



Features

- Linked to Thermo-Calc for thermodynamics
 - Has all modules in Thermo-Calc plus a Dictra module.
- Based on thermodynamics and mobility databases.
 - Quantitative description of multicomponent diffusion
 - Mobility for simple phases and *ordered B2 phase*
 - Diffusion through stoichiometric phases (new in Version 22)
- Written in FORTRAN, using finite difference method, runs on most plattforms.
- Geometry: One dimensional
- Three types of phase interface
 - Local equilibrium
 - Finite interface mobility
 - Interfacial energy

Atomistic Treatment of Diffusion

- For crystalline phases it's usually believed that diffusion occurs through a vacancy exchange mechanism.
- Assuming that there is a random distribution of vacant sites and that the number of vacancies is everywhere adjusted to equilibrium, it's possible to derive the following expression for the flux of k in a lattice-fixed frame of reference:

$$J_k^L = -c_k y_{va} M_{kva} \nabla \mu_k$$

- where M_{kva} is some kinetic factor which gives the rate of exchange if there is a vacancy adjacent to a k atom.

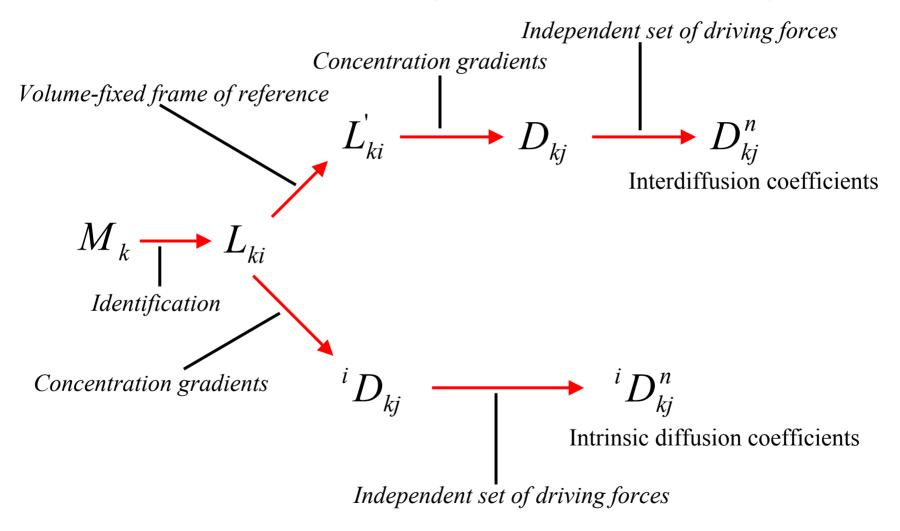
Phenomenological equations
$$J_{k}^{L} = -\sum_{i=1}^{n} L_{ki} \frac{\partial \mu_{i}}{\partial z} - L_{1T} \frac{\partial T}{\partial z} - L_{1P} \frac{\partial P}{\partial z} - L_{1\phi} \frac{\partial \phi}{\partial z}$$

They are called phenomenological since they steam from no model, but from the observed conditions of equilibrium.

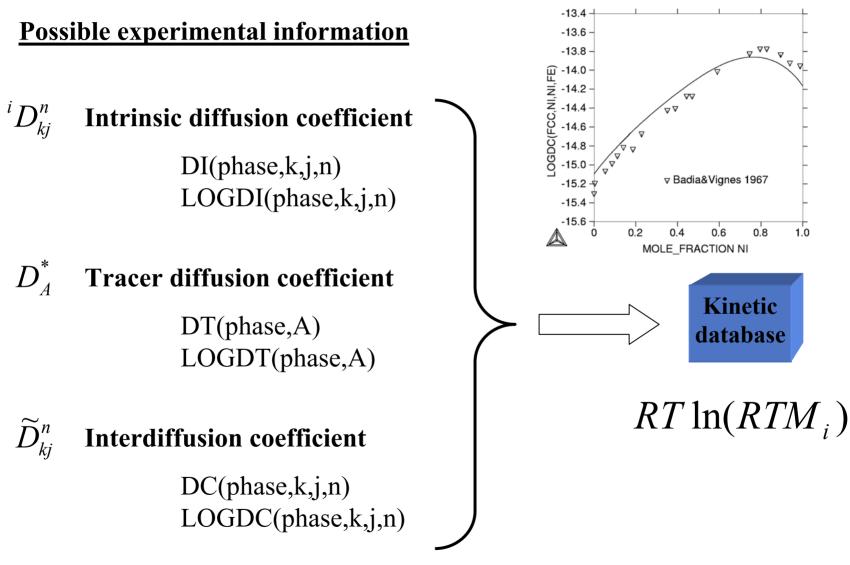
If we choose to consider an isothermal, isobaric and isopotential system we have:

$$J_{k}^{L} = -\sum_{i=1}^{n} L_{ki} \frac{\partial \mu_{i}}{\partial z} \qquad \left(J_{k}^{L} = -L_{kk} \frac{\partial \mu_{k}}{\partial z}\right)$$

From Mobility to Diffusibity



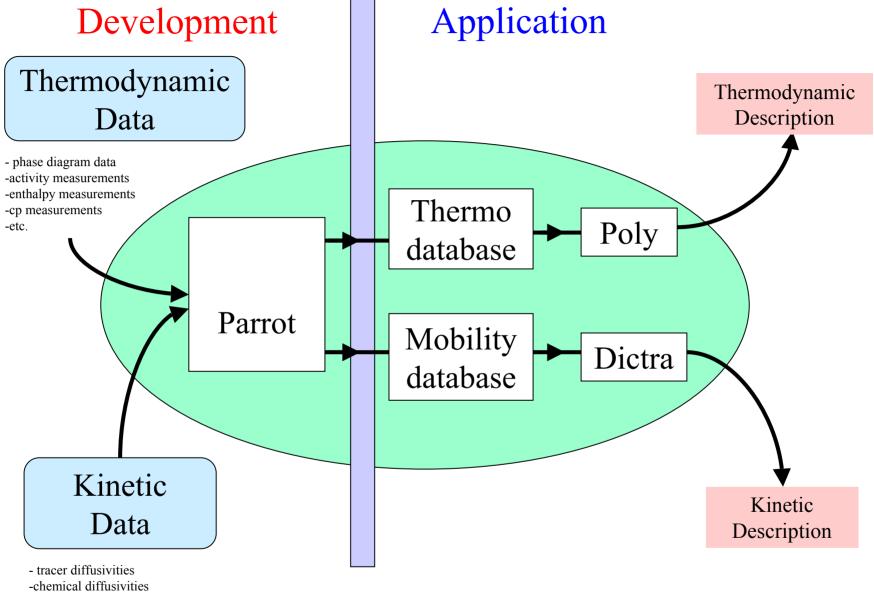
Kinetic databases



Modeling

$$M_{i} = \frac{1}{RT} \exp\left(\frac{\Delta G_{i}}{RT}\right) \qquad \Delta G_{i} = -Q_{i} + RT \ln(M_{i}^{0})$$

$$\begin{split} \Delta G_i &= \sum_j \sum_m y_j^I y_m^{II} \Delta G_i^{j:m} + \sum_j \sum_{k>j} \sum_m y_j^I y_k^{II} y_m^{II} \Delta G_i^{j,k:m} \\ &+ \sum_j \sum_n \sum_{m>n} y_j^I y_n^{II} y_m^{II} \Delta G_i^{j:n,m} \end{split}$$



⁻ etc.

Modeling Steps

- Understand the system and select experimental data
 - Search and read all references
- Create a setup file for the system
 - Define parameters
- Create an experimental data file
 - Make a pop file (or called dop file)
- Evaluate mobility model parameters
 - Use Parrot
- Compare experimental and calculated results
 - Use Dictra module
- Make a mobility database
 - add to an existing database

Mobility Databases

MOB2

The most general diffusion database

Could be used for Steels/Fe_alloys, Ni-alloys, Al-based alloys, and more.

75 Elements: Ag, Al, Am, As, Au, B, Ba, Be, Bi, C, Ca, Cd, Co, Cr, Cs, Cu, Dy, Er, Fe, Ga, Gd, Ge, Hf, Hg, Ho, In, Ir, K, La, Li, Lu, Mg, Mn, Mo, N, Na, Nb, Nd, Ni, Np, Os, P, Pa, Pb, Pd, Pr, Pt, Pu, Rb, Re, Rh, Ru, S, Sb, Sc, Se, Si, Sm, Sn, Sr, Ta, Tb, Tc, Te, Th, Ti, TI, Tm, U, V, W, Y, Yb, Zn and Zr

Phases with diffusion data: BCC_A2, CEMENTITE, FCC_A1, FE4N, HCP_A3, LIQUID

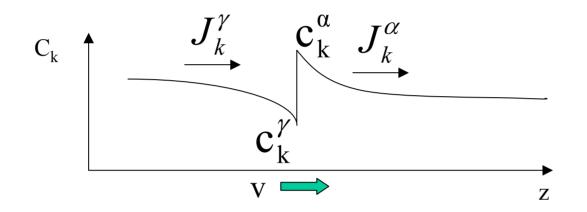
Assessed data for the following binary systems BCC_A2: C-Fe, C-Cr, Cr-Fe, Cr-N, Cr-Ni, Fe-N, Fe-Ni FCC_A1: Al-Cr, Al-Ni, C-Fe, C-Ni, Cr-Fe, Cr-Ni, Fe-N, Fe-Ni, Fe-Si HCP_A3: C-Fe, Fe-N FE4N: C-Fe, Fe-N

Assessed data for the following ternary systems

BCC_A2: C-Cr-Fe FCC_A1: Al-Cr-Ni, C-Cr-Fe, C-Fe-Ni

Assessed data for the following higher order systems BCC_A2: C-Cr-Fe-N-Ni FCC_A1: C-Cr-Fe-Ni

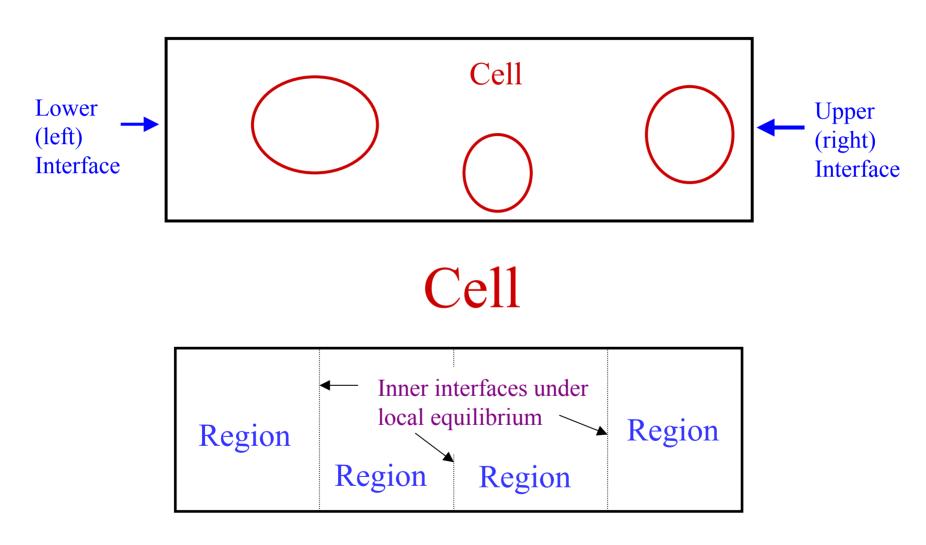
Sharp Interface



Unknowns: Tie-line, specified by n-2 a_i or μ_i Velocity of phase boundary, v

Equations: n-1 flux-balance equations, $v(c_k^{\alpha} - c_k^{\gamma}) = J_k^{\alpha} - J_k^{\gamma}$ Software: Dictra

System

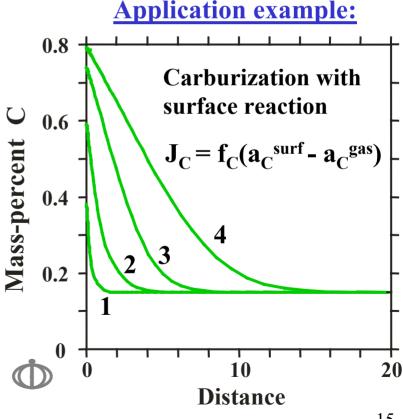


Boundary conditions in DICTRA

- Boundary Conditions can be specified as functions of time, temperature and pressure.
- Different functions may be used in different time intervals.

Example of conditions are:

- Closed system (default)
- Sate variable expressions
- Flux conditions
- Mixed conditions



ENTER GEOMETRICAL EXPONENT Command that determines the geometry of the system. The program handles problems that can be reduced to 1-dimensional geometries. Default value is zero, i.e. planar geometry.



Exponent Geometry

0 Planar geometry.

This corresponds to an infinitely wide plate of a certain thickness.

1 Cylindrical geometry.

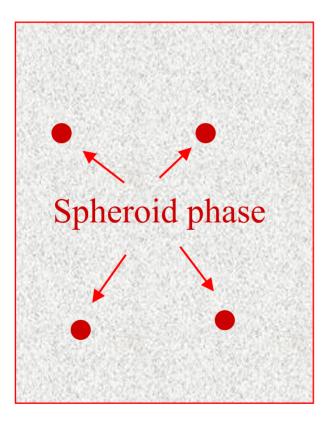
This corresponds to an infinitely long cylinder of a certain radius.

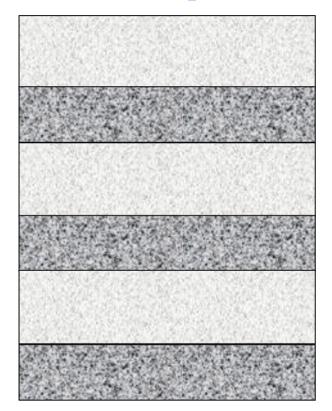
2 Spherical geometry. A sphere with a certain radius.



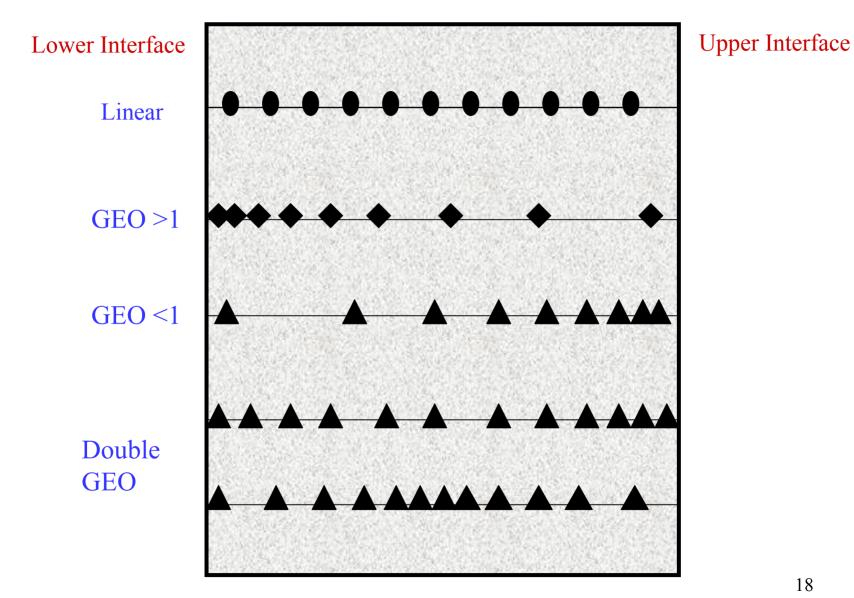
Matrix phase

Lamella phase



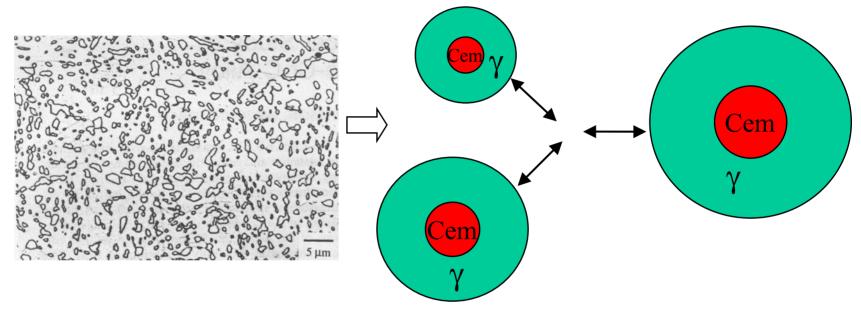


Grid points



Cementite dissolution in a Fe–Cr-C alloy Cell calculation with size-distribution

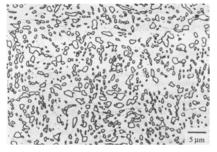
Dissolution of cementite at 910C (or 1183K):

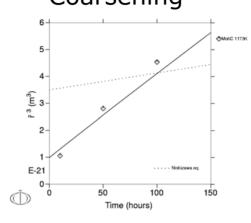


Instead of assuming an average particles size as in previous example, we will perform the simulation for a particle distribution using three cells of different sizes.

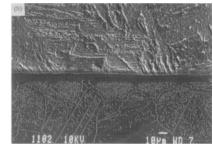
Some DICTRA applications

Carbide dissolution

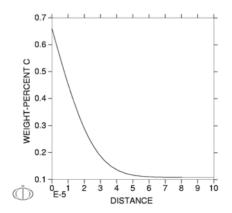


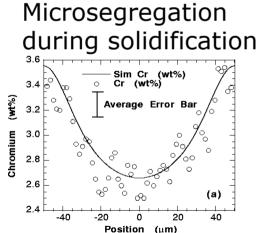


Carbon diffusion in a weld between two steels



Carburizing





Lippard et al. Metall. Mater. Trans. B 1998

More DICTRA applications

- Homogenisation of alloys
- Carburization and decarburization
- Carburization of alloys
- Nitriding of steels
- Diffusion during sintering of cemented carbides
- Nitrocarburizing of steels
- Austenite/ferrite diffusional transformations in steels
- Growth or dissolution of individual particles

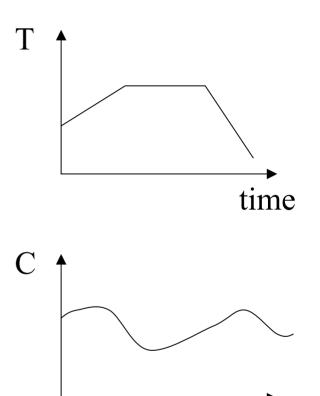
- Transient liquid phase bonding of alloys
- Calculation of TTT-diagrams
- Interdiffusion between coating/substrate
- •Coarsening of a particle distribution
- Gradient sintering of cemented carbides
- Growth of pearlite in alloyed steels
- Sigma phase precipitation in stainless steel
- Post weld heat treatment of welds between dissimilar materials

Input of T and c

• Temperature (T) can be entered as a function of time

Many different functions can be used (+, -, *, **, SQRT(X), EXP(X),LOG(X),SIN(X))

- Concentration can be entered as a function of distance or read from a file
- Many functions e.g. error-functions (erf(x)) and heavy-side step functions (hs(x))



Distance (X)

Summary

- DICTRA integrates the thermodynamic database and atomic mobility databases and solves one-dimensional diffusion equations in multicomponent systems.
- DICTRA can be used to develop atomic mobility databases using various experimental diffusion data.
- DICTRA can treat interface migrations with finite interfacial mobility and diffusion through stoichiometric phases.
- Any comments/questions, contact liu@matse.psu.edu.